# Photoelectrochemical  $CO<sub>2</sub>$  Reduction into Syngas with the Metal/ Oxide Interface

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**S** [Supporting Information](#page-6-0)

ABSTRACT: Photoelectrochemical (PEC) reduction of  $CO<sub>2</sub>$  with  $H<sub>2</sub>O$  not only provides an opportunity for reducing net  $CO<sub>2</sub>$  emissions but also produces valueadded chemical feedstocks and fuels. Syngas, a mixture of  $CO$  and  $H<sub>2</sub>$ , is a key feedstock for the production of methanol and other commodity hydrocarbons in industry. However, it is challenging to achieve efficient and stable PEC  $CO<sub>2</sub>$  reduction into syngas with controlled composition owing to the difficulties associated with the chemical inertness of  $CO<sub>2</sub>$  and complex reaction network of  $CO<sub>2</sub>$  conversion. Herein, by employing a metal/oxide interface to spontaneously activate  $CO<sub>2</sub>$  molecule and stabilize the key reaction intermediates, we report a benchmarking solar-to-syngas efficiency of 0.87% and a high turnover number of 24 800, as well as a desirable high stability of 10 h. Moreover, the  $CO/H<sub>2</sub>$  ratios in the composition can be tuned in a



wide range between 4:1 and 1:6 with a total unity Faradaic efficiency. On the basis of experimental measurements and theoretical calculations, we present that the metal/oxide interface provides multifunctional catalytic sites with complementary chemical properties for CO<sub>2</sub> activation and conversion, leading to a unique pathway that is inaccessible with the individual components. The present approach opens new opportunities to rationally develop high-performance PEC systems for selective CO<sub>2</sub> reduction into valuable carbon-based chemicals and fuels.

# **■ INTRODUCTION**

The development of solar-powered  $CO<sub>2</sub>$  reduction with  $H<sub>2</sub>O$ holds the promise to mitigate greenhouse gas  $(CO<sub>2</sub>)$  emission into the atmosphere, while simultaneously converting renewable solar energy into storable value-added chemicals and fuels.[1](#page-6-0)−[5](#page-6-0) The photoelectrochemical (PEC) route, which combines light harvesting photovoltaic and electrochemical components into a monolithically integrated device, has received considerable attention for application in  $CO<sub>2</sub>$ reduction recently.<sup>[6](#page-6-0)−[9](#page-6-0)</sup> Among the wide variety of  $CO_2$ reduction products, CO is a gaseous product that requires only two proton−electron transfers, and thus a kinetically feasible choice compared to  $CH<sub>3</sub>OH$  and  $CH<sub>4</sub>$ , which require six and eight proton−electron transfers to form one molecule, respectively.[10](#page-6-0),[11](#page-6-0) Moreover, CO is an important bulk chemical to form methanol and other commodity hydrocarbons via syngas intermediate, a mixture of CO and  $H_2$ , by using wellestablished standard industrial processes such as Fischer− Tropsch technology.[12](#page-6-0)−[15](#page-6-0) These attributes, together with the almost inevitable  $H_2$  evolution in aqueous PEC cell can be valorized to produce syngas mixtures, rendering the proposed syngas production from  $CO<sub>2</sub>$  and  $H<sub>2</sub>O$  conversion a technologically and economically viable pathway to leverage

established commercial processes for liquid fuels synthesis. Providing different  $CO/H_2$  ratio in syngas mixtures can also be used for different downstream products (e.g., 1:3, 1:2 and 1:1 for methane, methanol and oxo-alcohols, respectively).<sup>[16](#page-6-0)−[18](#page-6-0)</sup> Therefore, the syngas route would allow a flexible platform for integration with a wide window of catalytic systems in a broad  $CO<sub>2</sub>$ -recycling scheme without the strict requirement of suppression of  $H_2$  evolution reaction.<sup>[19](#page-6-0)–[31](#page-7-0)</sup>

To date, various semiconductor photocathodes, including p-Si,<sup>[32](#page-7-0)–[34](#page-7-0)</sup> ZnTe,<sup>35,36</sup> CdTe,<sup>[37](#page-7-0)</sup> p-InP,<sup>37</sup> Cu<sub>2</sub>O<sup>[38,39](#page-7-0)</sup> and p-NiO,<sup>[40,41](#page-7-0)</sup> have been investigated for PEC  $CO<sub>2</sub>$  reduction into  $CO$ , usually in conjunction with a molecular metal-complex or metal cocatalyst (e.g., Au, Ag and derivatives) to realize selective CO production. However, it remains challenging to develop efficient and stable PEC catalytic system that can activate inert  $CO<sub>2</sub>$  molecule at low overpotential or even spontaneously, and selectively produce syngas with controlled composition in a wide range to meet different downstream products. It has been reported that pure metal catalyst with a simple monofunctional site usually has a weak interaction with  $CO<sub>2</sub>$  molecule and

Received: March 19, 2018 Published: June 15, 2018



Figure 1. Characterization of Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si sample. (a) Schematic illustration of the structure, (b) 45°-tilted SEM image shows GaN nanowire growth vertically on the Si substrate, (c) TEM image illustrates Pt nanoparticles distributed uniformly on the GaN nanowire surface, (d) HRTEM image, and EDX analysis of the center (e) and edge region (f) indicates the coating of GaN nanowire and Pt nanoparticles with ultrathin  $TiO<sub>2</sub>$  layer. The Cu peaks in EDX arise from the TEM sample grid.

cannot provide multiple sites for stabilizing the key reaction intermediates with optimal binding strength, which leads to impractically high overpotential and low catalytic efficiency/ stabilty. $42,43$  $42,43$  $42,43$  Here we show that efficient and stable syngas production with tunable composition from PEC  $CO<sub>2</sub>$  reduction can be achieved with the introduction of a completely different type of sites: metal/oxide interface. On the basis of experiments and first-principles theoretical calculations, it is found that the metal/oxide interface can spontaneously activate  $CO<sub>2</sub>$  and stabilize the key reaction intermediates for facilitating CO production. The intimate metal/oxide interface provides the multifunctional combination of metal and oxide catalytic sites with complementary chemical properties, which opens new reaction channels that are not possible with the individual components. The versatility of using metal/oxide interface is demonstrated by the combination of different metals (Pt and Pd) and oxides (TiO<sub>2</sub> and ZnO). Remarkably, although pristine metal catalytically favors the proton reduction to evolve  $H<sub>2</sub>$ , the coverage of metal with oxide to form metal/oxide interface shows preferential activity for  $CO_2$  reduction over  $H_2$  evolution. As an example, by rationally integrating the  $Pt/TiO<sub>2</sub>$  cocatalyst with the strong light harvesting of p-n Si junction and efficient electron extraction effect of GaN nanowire arrays  $(Pt-TiO<sub>2</sub>/$ GaN/n<sup>+</sup> -p Si), a record half-cell solar-to-syngas (STS) efficiency of 0.87% and a benchmark turnover number (TON) of 24 800 have been achieved in an aqueous PEC system. The durability of the PEC system for highly stable syngas production of 10 h has been demonstrated as well.

# ■ RESULTS AND DISCUSSION

Design and Synthesis of Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si. We choose Pt-TiO<sub>2</sub> as an example to demonstrate the validity of metal/oxide interface for PEC  $CO<sub>2</sub>$  reduction. GaN nanowire

on p-n Si junction was selected as the platform to load the Pt-TiO<sub>2</sub> system. The schematic design of Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si is illustrated in Figure 1a. The sample was prepared in two major steps. First, GaN nanowire arrays were grown on p-n Si wafer by plasma-assisted molecular beam epitaxy (see the [Supporting](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf) [Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf).[31,44](#page-7-0) Such a structure takes advantage of the strong light absorption capability of Si (bandgap of 1.1 eV) and efficient electron extraction effect as well as large surface area provided by GaN nanowires. Moreover, the light absorption and catalytic reaction sites are decoupled spatially in the structure, providing an ideal platform to investigate the effect of cocatalysts on the catalytic performance without affecting the optical properties. Second, Pt nanoparticles and  $TiO<sub>2</sub>$  ultrathin layer were deposited on GaN nanowires surface in sequential order using photodeposition and atomic-layer deposition (ALD) process, respectively (see the [Supporting Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf). The intimate  $Pt/TiO<sub>2</sub>$  interface provides multiple sites and unique channels that facilitate the  $CO<sub>2</sub>$  activation and reaction pathways for syngas production.

Characterization of Structure. The morphology and chemical component of the Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si heterostructures were studied using scanning electron microscopy (SEM), transmission electron microscopy (TEM), energydispersive X-ray spectroscopy (EDX) and inductively coupled plasma−atomic emission spectroscopy (ICP-AES) analysis. The cross-sectional SEM image (Figure 1b) shows that the GaN nanowires are aligned vertically to the Si substrate with an average diameter of ∼50 nm (±15 nm) and height of 250 nm (±50 nm). TEM image in Figure 1c reveals that Pt nanoparticles of 2−3 nm size are uniformly deposited on the GaN nanowires surface. High-resolution TEM (HRTEM) image (Figure 1d), along with EDX analysis in the center and edge regions of nanowire (Figures 1e and 1f, respectively),

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Figure 2. (a) Faradaic efficiencies for CO (gray bars) and  $H_2$  (red bars), and solar-to-syngas efficiency of Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si photocathode as a function of potential in CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> solution (pH 7.5). (b) Chronoamperometry data and FEs for CO and H<sub>2</sub> of Pt-TiO<sub>2</sub>/GaN/ n<sup>+</sup>-p Si photocathode at +0.27 V vs RHE. The dashed lines denote cleaning of photoelectrode and purging of the PEC cell with CO<sub>2</sub>. (c) J−V curves of bare GaN/n<sup>+</sup>-p Si, GaN/n<sup>+</sup>-p Si with individual Pt or TiO<sub>2</sub> cocatalyst, and Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si. (d) Faradaic efficiencies for CO at +0.27 V vs RHE. The FEs for CO of GaN/n<sup>+</sup>-p Si and TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si photocathodes were measured at −0.33 V vs RHE due to the negligible photocurrent at an applied positive potential. The light illumination is ∼8 suns.

indicates the coating of GaN nanowire with ultrathin  $TiO<sub>2</sub>$ layer. The TiO<sub>2</sub> layer is amorphous and has a thickness of  $\sim$ 1 nm, which corresponds to 18 ALD cycles of  $TiO<sub>2</sub>$  deposition. The lattice spacings of 0.22 and 0.26 nm corresponds to the (111) facet of Pt and (002) lattice plane of GaN, indicating the preferred nanowire growth along ⟨0001⟩ direction (c-axis). The loading amounts of Pt and Ti in Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si were determined to be 4.9 and 48.3 nmol  $cm^{-2}$ , respectively, by using ICP-AES analysis.

Photoelectrochemical Properties. PEC performance of the sample was investigated in  $CO_2$ -saturated 0.5 M KHCO<sub>3</sub> solution (pH 7.5) under 300 W xenon lamp irradiation (800 mW cm<sup>−</sup><sup>2</sup> , ∼8 suns) in a conventional three-electrode cell. To reveal the interaction of photocathode with  $CO<sub>2</sub>$ , the current− potential  $(J-V)$  curves of Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si in a CO<sub>2</sub> or Ar-saturated electrolyte was compared (Figure S1, [Supporting](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf) [Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf). There is a large enhancement in the photocurrent generation under  $CO<sub>2</sub>$  atmosphere compared to that of Ar atmosphere, indicating an interaction between the electrode surface and  $CO<sub>2</sub>$  molecule for  $CO<sub>2</sub>$  reduction. Figure 2a shows the Faradaic efficiencies (FEs) for CO and  $H_2$  on Pt-TiO<sub>2</sub>/  $\text{GaN/n}^+$ -p Si at applied potential between +0.47 V and +0.07 V vs reversible hydrogen electrode (RHE) in  $CO_2$ -saturated electrolyte (hereafter, all the potentials are referenced to the RHE unless otherwise specified). The corresponding chronoamperometry data at different applied potentials are shown in Figure S2 ([Supporting Information](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf)). At an applied potential of +0.47 V, the photocathode exhibited a high CO FE of 78%, indicating the major extracted photogenerated electrons were used for selectively  $CO_2$ -to-CO conversion at the catalyst surface. By tuning the potential from +0.47 V to +0.07 V, the  $CO/H<sub>2</sub>$  ratio can be tuned in a large range between 4:1 and 1:6. At + 0.27 V, a  $CO/H_2$  ratio of 1:2 is obtained, which is a desirable composition of syngas mixtures for methanol synthesis and Fischer–Tropsch hydrocarbon formation.<sup>[45](#page-7-0)</sup> The

decreased CO FE at a more negative potential than +0.37 V is mainly due to the limited  $CO<sub>2</sub>$  mass transport in the electrolyte at high CO generation rate.  $46,47$  $46,47$  $46,47$  The kinetic limitation was evidenced by the saturated current density for CO generation in the high applied bias region (Figure S3a, [Supporting](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf) [Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf). In addition, different Tafel slopes for the  $CO<sub>2</sub>$ reduction and  $H_2$  evolution reactions could lead to the abovementioned bias-dependent reaction selectivity. To evaluate their contribution, the Tafel plots for CO and  $H_2$  evolution were drawn by using the corresponding partial current density, as shown in Figure S3b in the [Supporting Information.](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf) The Tafel slopes were calculated by using data points more positive than +0.37 V, as the slope increases dramatically at more negative potentials due to the mass-transport limitations.<sup>[48](#page-7-0),[49](#page-7-0)</sup> It was found that the Tafel slopes for CO and  $H_2$  evolution were 386 and 119 mV dec<sup>-1</sup>, respectively. The different Tafel slopes result in the bias-dependent reaction selectivity largely in the low bias region. At all the applied potentials, a total FE of 97  $\pm$ 8% was obtained for the cogeneration of CO and  $H<sub>2</sub>$ , with no appreciable amount of other gas products detected by gas chromatograph (GC) and liquid products (e.g., HCOOH and  $CH<sub>3</sub>OH$ ) analyzed by nuclear magnetic resonance (NMR) spectroscopy. To demonstrate that the generated CO from  $CO<sub>2</sub>$  reduction, isotopic experiment using  $^{13}CO<sub>2</sub>$  was conducted. The signal at  $m/z = 29$  assigned to <sup>13</sup>CO appeared in the gas chromatography−mass spectrometry analysis (Figure S4, [Supporting Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf), indicating the CO product is formed from the reduction of  $CO<sub>2</sub>$ .

A highlight of our system is the highly positive onset potential of +0.47 V (underpotential of 580 mV to the  $CO<sub>2</sub>/$ CO equilibrium potential at −0.11 V) for producing high CO FE of 78% in an aqueous PEC cell. The reported onset potentials, FEs and solar energy conversion efficiency of different photocathodes for CO production in an aqueous PEC cell are compared (Table S1, [Supporting Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf). Among various reported photocathodes, our system features the lowest onset potential, which is 170 mV positive shifted compared with the best value reported in the literature.<sup>[36](#page-7-0)</sup> The extremely low onset potential of our photocathode is attributed to coupling effects including strong light harvesting of p-n Si junction, efficient electron extraction of GaN nanowire arrays, and extremely fast syngas production kinetics on Pt-TiO<sub>2</sub> dual cocatalysts. The STS efficiencies of our system at different applied potentials were calculated according to the measured photocurrent density and FEs for CO and  $H_2$  ([eq 1,](#page-4-0) [Supporting](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf) [Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf). As shown in [Figure 2](#page-2-0)a, at +0.17 V, the STS efficiency reached 0.87%, which greatly outperforms other reported photocathodes and sets a new benchmark reported to date (Table S1, [Supporting Information](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf)).

The durability of Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si photocathode was investigated at a constant potential of +0.27 V by five consecutive runs with each run of 2 h ([Figure 2](#page-2-0)b). After each cycle, the products of  $CO$  and  $H<sub>2</sub>$  were analyzed by  $GC$ , the electrode was thoroughly cleaned by deionized water and the PEC cell was purged by  $CO<sub>2</sub>$  for 20 min. During the five runs of 10 h operation, the electrode showed similar behavior in terms of photocurrent density and product selectivity, indicating the high stability of the sample during the syngas production process. The initial decrease of high photocurrent density in each run is likely due to the limited mass transfer of reactants or products at high reaction rates, which can be recovered in the next run after the cleaning of photoelectrode surface. The CO/  $H_2$  ratio in the products was kept nearly 1:2 during the five cycles of operation, which is a desirable syngas composition for synthetizing downstream products including methanol and liquid hydrocarbons.<sup>[45](#page-7-0)</sup> In addition, the SEM, TEM, and XPS analysis of Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si photocathode after the PEC reaction were performed, as shown in Figure S5 [\(Supporting](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf) [Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf). No appreciable change of GaN nanowires and Pt-TiO<sub>2</sub> catalysts were found. The total turnover number (TON), defined as the ratio of the total amount of syngas evolved (264  $\mu$ mol) to the amount of Pt-TiO<sub>2</sub> catalyst (10.64 nmol, calculated from the catalyst loadings and electrode sample area of 0.2  $\rm cm^2)$ , reached 24 800, which is at least 1 or 2 orders of magnitude higher than previously reported values for syngas or CO formation from PEC or photochemical  $CO<sub>2</sub>$ reduction.[31,50](#page-7-0)−[53](#page-7-0)

Investigation of the Catalytic Mechanisms. To understand the underlying catalytic mechanism and the role of basic components for the PEC performance of the Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>p Si photocathode, we conducted a series of control experiments. [Figure 2](#page-2-0)c shows the comparison of LSV curves for bare GaN/n<sup>+</sup>-p Si, GaN/n<sup>+</sup>-p Si with individual Pt or TiO<sub>2</sub> cocatalyst, and  $Pt-TiO_2/GaN/n^+$ -p Si. The bare  $GaN/n^+$ -p Si displays a poor PEC performance with a negligible photocurrent density and highly negative onset potential. The loading of Pt cocatalyst can greatly improve the PEC performance with an onset potential of about +0.47 V and photocurrent density of ~50 mA cm<sup>-2</sup> at −0.33 V, while TiO<sub>2</sub> alone shows a small photocurrent density of 5 mA cm<sup>−</sup><sup>2</sup> at −0.33 V. Compared to bare Pt, significantly higher photocurrent density of ∼120 mA cm<sup>-2</sup> at -0.33 V is attained when Pt and TiO<sub>2</sub> are loaded simultaneously. It is proposed that the formation of intimate  $Pt/TiO<sub>2</sub>$  interface stabilizes the reaction intermediates and reduces the activation barrier for syngas production, which are validated by theoretical calculations discussed below. In addition, the ultrathin  $TiO<sub>2</sub>$  overlayer may passivate the nanowire surface states and reduce the probability of charge

carrier recombination at the surface.<sup>[54](#page-7-0),[55](#page-7-0)</sup> It is also supposed that  $Pt/TiO<sub>2</sub>$  interface is more resistant to CO poisoning than Pt alone as shown in thermochemical catalysis,<sup>[56](#page-7-0)–[58](#page-7-0)</sup> which could contribute to the enhanced syngas production on metal/oxide interface. [Figure 2](#page-2-0)d shows the comparison of FEs of CO for the four samples. Besides CO product, the remaining balance of photocurrent drives  $H_2$  evolution from proton reduction. It is shown that CO FEs are very low on bare GaN/n<sup>+</sup>-p Si, and with individual Pt or TiO<sub>2</sub> cocatalyst (1.7%, 2% and 5.6%, respectively). In contrast, the CO formation selectivity increases greatly to 32% by loading Pt-TiO<sub>2</sub> dual cocatalyst, indicating a synergetic effect between Pt and  $TiO<sub>2</sub>$ . We attribute the synergy to the strong interaction at the intimate metal/ oxide interface, which provides the multifunctional adsorption/ reaction sites for  $CO<sub>2</sub>$  activation and conversion. It is worth mentioning that there is an optimized thickness of  $\sim$ 1 nm TiO<sub>2</sub> for maximum catalytic activity and CO selectivity (Figure S6, [Supporting Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf). Very thin  $TiO<sub>2</sub>$  deposition yields less interfacial reactive sites, while increasing the  $TiO<sub>2</sub>$  thickness over 1 nm resulted in limited mass transport of reactants to the interfacial sites and large tunneling resistance to charge carrier transport associated with thick  $TiO<sub>2</sub>$  layer.<sup>[59,60](#page-7-0)</sup>

CO<sub>2</sub> Adsorption and Activation. To elucidate the role of metal/oxide interface for the conversion of  $CO<sub>2</sub>$  to  $CO$  from the fundamental atomic level, density functional theory (DFT) calculations were employed using  $Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/Pt(111)$  to describe the  $Pt/TiO<sub>2</sub>$  interface (see [Supporting Information](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf) for computational details). The hydroxylation of titania cluster  $(T_i, O_6H_6)$  was considered in the calculations to account for the effect of PEC  $CO<sub>2</sub>$  reduction conditions in an aqueous environment.<sup>[61,62](#page-7-0)</sup> The optimized geometry of Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/Pt-(111) is presented in Figure S7 ([Supporting Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf). As  $CO<sub>2</sub>$  adsorption and activation on catalyst surface is the initial and often the rate-determining step for the whole  $CO<sub>2</sub>$ reduction process,  $63,64$  $63,64$  $63,64$  we first investigate the CO<sub>2</sub> adsorption characteristics on Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/Pt(111) surface. The calculation of  $CO<sub>2</sub>$  adsorption on pristine Pt(111) was also performed as a comparison. [Figures 3](#page-4-0)a and [3b](#page-4-0) show the optimized configurations of  $CO<sub>2</sub>$  adsorption on the pristine  $Pt(111)$  and  $Ti_3O_6H_6/Pt(111)$  surface, respectively. It was found that  $CO_2$ retains the original linear configuration on pristine  $Pt(111)$ , similar to its isolated gas-phase state. In contrast, there are strong interactions between  $CO<sub>2</sub>$  molecule and the Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/ Pt(111) interface, with C atom strongly binding to the Pt atom underneath with a bond length of 2.02 Å and one O atom (O2) attaching to the Ti atom with a shorter bond length of 1.96 Å. Such a strong bonding between  $CO_2$  and  $Ti_3O_6H_6/Pt(111)$ interface results in a significant bending of  $CO<sub>2</sub>$  molecule from its originally linear form to an O−C−O angle of 125.02°, thus forming a tridentate configuration that facilitates its subsequent transformations.<sup>[64](#page-7-0)</sup> In addition, the strong interaction of  $CO<sub>2</sub>$ with the interface weakens the two C−O bonds of  $CO_2$ , leading to elongated C−O bonds (1.22 and 1.32 Å) from the original bond length of 1.18 Å in the isolated  $CO<sub>2</sub>$  molecule (Table S2, [Supporting Information](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf)). The weakened C−O bonds and the formed bent  $CO<sub>2</sub>$  configurations indicate a remarkable activation of  $CO<sub>2</sub>$  molecule upon chemisorption at the interface, which is in contrast with the negligible activation of  $CO<sub>2</sub>$  on pristine Pt(111). This result agrees well with the observations in the field of thermochemical catalysis that  $CO<sub>2</sub>$ transformation is greatly enhanced with metal/oxide interface as compared to that with pure metal.<sup>[65](#page-7-0)–[69](#page-7-0)</sup> It is worth noting that the  $CO<sub>2</sub>$  activation mechanism at metal/oxide interface has

<span id="page-4-0"></span>

Figure 3. Side views of optimized configurations of CO<sub>2</sub> adsorbed on the (a) Pt(111) surface and (b)  $Ti_3O_6H_6/Pt(111)$  surface. (c) Differential charge density of  $CO_2$  adsorbed at the Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/Pt(111) interface. Regions of yellow and blue indicate electronic charge gain and loss, respectively. Isosurface contours of electron density differences were drawn at  $\pm 0.002$  e/Bohr<sup>3</sup>. (d) Calculated free energy diagrams for  $CO_2$  reduction to CO on Pt(111) and Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/Pt(111) surfaces at 0 V vs RHE. The optimized structures for each step are also shown. To improve legibility, a break region was added from 0.25 to 3.75 on the Y axis due to the large energy barriers for the  $CO<sub>2</sub>$ reduction on Pt(111) surface. Pt: gray, Ti: blue, O: red, C: brown and H: white.

a certain degree of similarity to that reported on individual metal oxide (e.g.,  $TiO<sub>2</sub>$ ) with oxygen vacancies, in which one of the O atoms in  $CO<sub>2</sub>$  is coordinating to an undercoordinated Ti atom at the edge of the cluster (i.e., essentially an O vacancy).[70](#page-7-0)−[72](#page-7-0) For example, Somorjai and co-workers proposed a similar mechanism to explain carbonyl bond activation at the  $Pt/TiO<sub>2</sub>$  interface via oxygen vacancy on TiO<sub>2</sub> surface.<sup>[70](#page-7-0)</sup>

The energetics associated with  $CO<sub>2</sub>$  adsorption on Pt(111) and  $Ti_3O_6H_6/Pt(111)$  surfaces were also calculated and analyzed in terms of the adsorption energy  $(E_{ad})$  and deformation energy  $(E_{\text{def}}^{CO_2})$  (Table S2, [Supporting Informa](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf)[tion\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf). Here  $E_{\text{ad}}$  represents the net energy increased upon adsorption.  $E_{\text{def}}^{\text{CO}_{2}}$  denotes the energy change from the distortion of a linear  $CO_2$  molecule into a buckled configuration, correlating with the degree of  $CO_2$  activation.<sup>[73](#page-7-0)</sup> The  $E_{ad}$  and  $E_{\text{def}}^{\text{CO}_2}$  of  $\text{CO}_2$  adsorption at  $\text{Ti}_3\text{O}_6\text{H}_6/\text{Pt}(111)$  interface are −0.80 and 2.65 eV respectively, as compared with those of 4.44 and 0.01 eV on pristine Pt(111). The negative  $E_{ad}$  value implies the exothermic process of  $CO_2$  adsorption at  $Ti_3O_6H_6/Pt(111)$ interface, while positive  $E_{ad}$  value indicates the unfavorable  $CO_2$ adsorption on pristine  $Pt(111)$ . In addition, the large positive value of  $E_{\text{def}}^{\text{CO}_2}$  in the case of  $\text{Ti}_3\text{O}_6\text{H}_6/\text{Pt}(111)$  confirms that  $CO<sub>2</sub>$  is activated spontaneously at the interface, in strong contrast to the marginal value on pristine  $Pt(111)$ . Experimentally, the amount of  $CO<sub>2</sub>$  adsorption capacity over

 $Pt/GaN/n^{\text{+}}$ -p Si and  $Pt-TiO_2/GaN/n^{\text{+}}$ -p Si was tested by  $CO_2$ adsorption−desorption measurements (Figure S8, [Supporting](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf) [Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf). The CO<sub>2</sub> adsorption amount over Pt-TiO<sub>2</sub>/ GaN/n<sup>+</sup>-p Si was 1.91  $\mu$ mol cm<sup>-2</sup>, which was 7 times higher than that of Pt/GaN/n<sup>+</sup>-p Si (0.27  $\mu$ mol cm<sup>-2</sup>). As a comparison, the  $CO_2$  adsorption amount on plain GaN/n<sup>+</sup>-p Si was 0.24  $\mu$ mol cm<sup>-2</sup>, indicating the low propensity of Pt for  $CO<sub>2</sub>$  chemisorption. The combined experimental and theoretical results explain well the different behaviors in the PEC studies that pristine Pt does not favor  $CO<sub>2</sub>$  reduction, while the construction of  $Pt/TiO<sub>2</sub>$  interface shows greatly enhanced activity for CO<sub>2</sub> reduction.

To further investigate the detailed bonding interaction between  $CO_2$  and  $Ti_3O_6H_6/Pt(111)$  interface, the differential charge density (DCD) was examined, shown in Figure 3c. The yellow and blue regions indicate electronic charge accumulation and depletion, respectively. Strong electronic coupling between  $CO<sub>2</sub>$  and the interface was evidenced by the electron charge density redistribution around the interfacial region. Notable electron accumulation near the O2 atom in  $CO<sub>2</sub>$  and electron depletion around the neighboring Ti nucleus indicates an ioniclike Ti−O bonding, while the electron accumulation between Pt and C atoms suggests the formation of covalent Pt−C bonding. Overall, substantial electrons are transferred from the interface to  $CO<sub>2</sub>$  molecule, resulting in the formation of activated  $^*CO_2^-$  anion and eventually the enhanced  $CO_2$ reduction activity. Quantitative estimate of the electron transfer was studied by the Bader charge analysis (Table S2, [Supporting](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf) [Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf).<sup>[74,75](#page-7-0)</sup> It was found that  $CO_2$  attracted 0.684e from the substrate for  $CO_2$  adsorption at the  $Ti_3O_6H_6/Pt(111)$ interface, as compared to 0.0263e in the case of pristine Pt.

Reaction Energetics. Moreover, to gain insights into the selective CO evolution from  $CO<sub>2</sub>$  reduction at molecular level, DFT calculations were also performed to understand the reaction energetics of the  $CO<sub>2</sub> \rightarrow CO$  pathway. As suggested by previous studies,<sup>[76](#page-7-0)-[78](#page-7-0)</sup> we considered the following reaction steps:

 $CO<sub>2</sub>(g) + * + H<sup>+</sup>(aq) + e<sup>-</sup> \rightarrow *COOH$  (1)

\* COOH + H<sup>+</sup>(aq) + e<sup>-</sup> → \* CO + H<sub>2</sub>O(l) (2)

$$
^*CO \to CO(g) + ^* \tag{3}
$$

where a lone asterisk (\*) represents a surface adsorption site and \* symbol before a molecule denotes a surface-bound species. Figure 3d shows the calculated free energy diagram of  $CO<sub>2</sub>$  reduction on Pt(111) and Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/Pt(111). On pristine Pt(111), the first step of  $CO<sub>2</sub>$  activation to form \*COOH intermediate is highly endergonic with a free energy change  $(\Delta G)$  of 5.08 eV, which is the rate-limiting step for the whole  $CO_2$  reduction process. In contrast, on the Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/Pt(111) interface, \*COOH formation is exergonic owing to the strong binding to the interfacial sites, with C and O atoms in COOH binding to Pt(111) and Ti of Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>, respectively. Similarly, the strong binding and stabilization of \*CO intermediates were also observed with cooperative interactions with both metal and oxide in the interface, resulting in the facile formation of \*CO. The rate-limiting step in the  $Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/Pt(111)$  system is the CO desorption, but with a much smaller free energy change of 0.88 eV as compared to 5.08 eV on pristine Pt(111). This result suggests that there are sites of different nature with complementary chemical properties in the metal/oxide interface that work in synergy to facilitate the  $CO<sub>2</sub>$  reduction into

<span id="page-5-0"></span>

Figure 4. (a) XPS of Pt 4f of Pt/GaN/n<sup>+</sup>-p Si and Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si. (b) Electron localized function (ELF) of Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/Pt(111). The probability of finding electron pairs varies from 0 (blue color) to 1 (red color).



Figure 5. (a) Faradaic efficiencies for CO of Pd/GaN/n<sup>+</sup>-p Si, Pd-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si, Pt/GaN/n<sup>+</sup>-p Si and Pt-ZnO/GaN/n<sup>+</sup>-p Si. The measurements were performed at +0.27 V vs RHE for 100 min. (b) Calculated free energy diagrams for  $CO_2$  reduction to CO on Pd(111), Pt(111),  $Ti_3O_6H_6/Pd(111)$  and  $Zn_6O_6H_7/Pt(111)$  surfaces at 0 V vs RHE. The optimized structures for each step are also shown. To improve legibility, a break region was added from 0.25 to 2.75 on the Y axis due to the large energy barriers for the CO<sub>2</sub> reduction on Pd(111) and Pt(111) surface. Pd: pine green, Pt: gray, Ti: blue, Zn: purple, O: red, C: brown and H: white.

CO. In addition, the effects of the electrolyte and applied potential were considered in DFT calculations, similar conclusions were obtained. The calculation results and details are available in the [Supporting Information](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf) (Figures S9 and S10).

Considering that  $H_2$  product from proton reduction is the other important component in the syngas mixture besides CO, we also calculated free energy diagrams for  $H<sub>2</sub>$  evolution on pristine Pt(111) and Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/Pt(111), as shown in Figure S11a [\(Supporting Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf).  $Ti_3O_6H_6/Pt(111)$  shows a slightly lowered energy barrier than that on pristine  $Pt(111)$  by 0.06 eV. Considering that the uncertainty associated with DFT energy calculations is on the same order, $79$  the calculated energy barriers for hydrogen evolution reaction are comparable in the two cases. Recent studies have shown that the  $CO<sub>2</sub>$ reduction selectivity in competition with H<sub>2</sub> evolution is related to the difference between their two thermodynamic limiting potentials (denoted as  $U_{L}(CO_{2}) - U_{L}(H_{2}))$ . <sup>[80](#page-7-0)–[82](#page-7-0)</sup> Therefore, we have calculated the difference between limiting potentials for CO and  $H_2$  evolution. As seen in Figure S11b [\(Supporting](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf) [Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf),  $Ti_3O_6H_6/Pt(111)$  displays a significantly more positive value for  $U_L(CO_2) - U_L(H_2)$  than that on pristine Pt(111), indicating higher selectivity for  $CO<sub>2</sub>$  reduction to CO.

Strong Metal/Oxide Interaction. In addition to the important role of the metal/oxide interface in activating  $CO<sub>2</sub>$ and stabilizing the key reaction intermediates, the electronic modification of the Pt catalyst owing to the strong interaction between metal and oxide may also contribute to the selective  $CO_2$  reduction into  $CO$  on Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si photocathode. The electronic properties of Pt were evaluated using the peak energy of Pt 4f by X-ray photoelectron spectroscopy (XPS) analysis (Figure 4a). Compared to Pt/GaN/n<sup>+</sup> -p Si, a notable shift of ca. 0.5 eV to higher binding energy position was observed for Pt 4f in Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si. This shift is less pronounced than the binding energy difference between  $Pt^{0}$ and  $Pt^{2+}$  in PtO (ca. 1.5 eV),<sup>[83](#page-7-0)</sup> indicating the presence of electron deficient Pt species  $(Pr^{n+})$  in Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si. A significant electronic modification by strong metal/oxide interaction is likely responsible for this change of Pt oxidation state. $84-87$  $84-87$  $84-87$  To confirm the strong interaction between the metal and oxide, we computed the electron localized function (ELF) for  $Ti_3O_6H_6/Pt(111)$  system, as shown in Figure 4b. Topology analysis of ELF can effectively characterize the nature of different chemical bonding schemes,<sup>[88](#page-8-0)</sup> and has been used to estimate the degree of metal–support interactions.<sup>[89](#page-8-0)</sup> The ELF map of  $Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/Pt(111)$  shows that there is a significant electron redistribution in the regions between Pt and  $Ti_3O_6H_6$ , indicating strong interactions between them. The strong interactions can modify the electronic property of Pt and hence enhance CO<sub>2</sub> reduction.

Generalization to Other Metal/Oxide Systems. By understanding the  $CO<sub>2</sub>$  activation and conversion at the Pt/  $TiO<sub>2</sub>$  interface on an atomic level, we propose that the findings can be extended to other metal/oxide systems. To show the

## <span id="page-6-0"></span>**Journal of the American Chemical Society** Article Article Article Article Article Article Article Article Article

generality,  $Pd-TiO_2/GaN/n^+$ -p Si and  $Pt-ZnO/GaN/n^+$ -p Si were synthesized by varying either metal or oxide components (see the [Supporting Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf). The chemical components and structures were confirmed by TEM and EDX analysis (Figures S12−S13, [Supporting Information\)](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf). By using ICP-AES analysis, the loading amounts of Pd and Ti in Pd-TiO $_2/$  $\text{GaN}/\text{n}^+$ -p Si, Pt and Zn in Pt-ZnO/GaN/n<sup>+</sup>-p Si were determined to be 5.4 and 46.1, 4.7, and 39.1 nmol  $cm^{-2}$ , , respectively. The FEs of CO for Pd-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si and Pt-ZnO/GaN/n<sup>+</sup> -p Si were measured and compared with Pd/ GaN/n<sup>+</sup>-p Si and Pt/GaN/n<sup>+</sup>-p Si, respectively [\(Figure 5a](#page-5-0)). The CO FEs of Pd-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si and Pt-ZnO/GaN/n<sup>+</sup>-p Si are four and 11 times higher than that with individual metal cocatalysts, similar to the trend observed in Pt-TiO<sub>2</sub>/GaN/n<sup>+</sup>-p Si system. In addition, the free energy diagram of  $CO<sub>2</sub>$ reduction into CO were calculated to validate the experimental observations. Ti<sub>3</sub>O<sub>6</sub>H<sub>6</sub>/Pd(111) and Zn<sub>6</sub>O<sub>6</sub>H<sub>7</sub>/Pt(111) were used in the DFT calculations to describe the  $Pd/TiO<sub>2</sub>$  and Pt/ ZnO interface, respectively. As seen in [Figure 5](#page-5-0)b,  $Ti_3O_6H_6/$  $Pd(111)$  and  $Zn_6O_6H_7/Pt(111)$  shows a significantly lowered energy barrier than those on pristine Pd(111) and Pt(111). Similarly, it was found that the formation of  $^*CO$  from  $CO_2$ reduction via \*COOH intermediate is a facile downhill process in the presence of metal/oxide interface, while the first step of CO2 activation to form \*COOH is highly endergonic on pure metal surface. Although quantitative differences exist between different systems, similar qualitative trend indicates the critical role of metal/oxide interfaces in activating  $CO<sub>2</sub>$ , and stabilizing the key reaction intermediates for facilitating CO production. The present study could therefore provide a promising set of principles to enhance the  $CO<sub>2</sub>$  reduction performance by tuning the compositions and structures of metal/oxide interface.

# ■ CONCLUSIONS

In summary, we have demonstrated an efficient and stable PEC  $CO<sub>2</sub>$  reduction system for syngas production with controlled composition, by employing a metal/oxide interface to activate inert  $CO<sub>2</sub>$  molecule and stabilize the key reaction intermediates. Using  $Pt/TiO<sub>2</sub>$  as an example, a benchmarking solar-to-syngas efficiency of 0.87% and a high turnover number of 24 800 were achieved. Moreover, the PEC system exhibited highly stable syngas production in the 10 h duration test. On the basis of experimental measurements and theoretical calculations, it was found that the synergistic interactions at the metal/oxide interface provide unique reaction channels that structurally and electronically facilitate  $CO<sub>2</sub>$  conversion into CO. This work may open new opportunities for the design and development of high-performance photoelectrochemical systems for selective  $CO<sub>2</sub>$  reduction.

## ■ ASSOCIATED CONTENT

# **6** Supporting Information

The Supporting Information is available free of charge on the [ACS Publications website](http://pubs.acs.org) at DOI: [10.1021/jacs.8b03067](http://pubs.acs.org/doi/abs/10.1021/jacs.8b03067).

Experimental and computational methods, additional figures and tables ([PDF](http://pubs.acs.org/doi/suppl/10.1021/jacs.8b03067/suppl_file/ja8b03067_si_001.pdf))

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#### Author Contributions

 $\perp$ S.C., P. O., and P.G. contributed equally to this work. **Notes** 

The authors declare no competing financial interest.

## ■ ACKNOWLEDGMENTS

This work was supported by Emissions Reduction Alberta (ERA). Part of the work was conducted in the Microfabrication Facility at McGill University and the Michigan Center for Materials Characterization at the University of Michigan (NSF #DMR-0723032). We greatly acknowledge the financial support from McGill Engineering Doctoral Award and National Sciences and Engineering Research Council (NSERC) Discovery grant (grant # RGPIN-2017-05187). We also thank Supercomputer Consortium Laval UQAM McGill and Eastern Quebec for providing computing power.

## ■ REFERENCES

(1) Kumar, B.; Llorente, M.; Froehlich, J.; Dang, T.; Sathrum, A.; Kubiak, C. P. Annu. Rev. Phys. Chem. 2012, 63, 541.

(2) White, J. L.; Baruch, M. F.; Pander Iii, J. E.; Hu, Y.; Fortmeyer, I. C.; Park, J. E.; Zhang, T.; Liao, K.; Gu, J.; Yan, Y.; Shaw, T. W.; Abelev, E.; Bocarsly, A. B. Chem. Rev. 2015, 115, 12888.

(3) Habisreutinger, S. N.; Schmidt-Mende, L.; Stolarczyk, J. K. Angew. Chem., Int. Ed. 2013, 52, 7372.

(4) Tu, W. G.; Zhou, Y.; Zou, Z. G. Adv. Mater. 2014, 26, 4607.

(5) Xie, S. J.; Zhang, Q. H.; Liu, G. D.; Wang, Y. Chem. Commun. 2016, 52, 35.

(6) Wang, P.; Wang, S.; Wang, H.; Wu, Z.; Wang, L. Part. Part. Syst. Char. 2018, 35, 1700371.

(7) Pang, H.; Masuda, T.; Ye, J. H. Chem. - Asian J. 2018, 13, 127.

(8) Zhang, N.; Long, R.; Gao, C.; Xiong, Y. Sci. China Mater. 2018, 61, 771.

(9) Kalamaras, E.; Maroto-Valer, M.; Shao, M.; Xuan, J.; Wang, H. Catal. Today 2018, [DOI: 10.1016/j.cattod.2018.02.045](http://dx.doi.org/10.1016/j.cattod.2018.02.045).

(10) Vesborg, P. C. K.; Seger, B. Chem. Mater. 2016, 28, 8844.

- (11) Karatairi, E.; Miller, J. E. MRS Bull. 2017, 42, 878.
- (12) Bell, A. T. Catal. Rev.: Sci. Eng. 1981, 23, 203.

(13) Waugh, K. C. Catal. Today 1992, 15, 51.

(14) Dry, M. E. Catal. Today 2002, 71, 227.

(15) Khodakov, A. Y.; Chu, W.; Fongarland, P. Chem. Rev. 2007, 107, 1692.

(16) Wenzel, M.; Rihko-Struckmann, L.; Sundmacher, K. AIChE J. 2017, 63, 15.

(17) Foit, S. R.; Vinke, I. C.; de Haart, L. G. J.; Eichel, R. A. Angew. Chem., Int. Ed. 2017, 56, 5402.

(18) Hernandez, S.; Farkhondehfal, M. A.; Sastre, F.; Makkee, M.; Saracco, G.; Russo, N. Green Chem. 2017, 19, 2326.

(19) Ross, M. B.; Dinh, C. T.; Li, Y.; Kim, D.; De Luna, P.; Sargent, E. H.; Yang, P. D. J. Am. Chem. Soc. 2017, 139, 9359.

(20) Sheng, W. C.; Kattel, S.; Yao, S. Y.; Yan, B. H.; Liang, Z. X.; Hawxhurst, C. J.; Wu, Q. Y.; Chen, J. G. G. Energy Environ. Sci. 2017, 10, 1180.

(21) Xu, J. Q.; Li, X. D.; Liu, W.; Sun, Y. F.; Ju, Z. Y.; Yao, T.; Wang, C. M.; Ju, H. X.; Zhu, J. F.; Wei, S. Q.; Xie, Y. Angew. Chem., Int. Ed. 2017, 56, 9121.

(22) Guo, S. J.; Zhao, S. Q.; Wu, X. Q.; Li, H.; Zhou, Y. J.; Zhu, C.; Yang, N. J.; Jiang, X.; Gao, J.; Bai, L.; Liu, Y.; Lifshitz, Y.; Lee, S. T.; Kang, Z. H. Nat. Commun. 2017, 8, 1828.

(23) He, R.; Zhang, A.; Ding, Y. L.; Kong, T. Y.; Xiao, Q.; Li, H. L.; Liu, Y.; Zeng, J. Adv. Mater. 2018, 30, 1705872.

### <span id="page-7-0"></span>**Journal of the American Chemical Society** Article American Chemical Society Article American Chemical Society Article

- (24) Furler, P.; Scheffe, J. R.; Steinfeld, A. Energy Environ. Sci. 2012, 5, 6098.
- (25) Kumar, B.; Smieja, J. M.; Sasayama, A. F.; Kubiak, C. P. Chem. Commun. 2012, 48, 272.
- (26) Kang, P.; Chen, Z. F.; Nayak, A.; Zhang, S.; Meyer, T. J. Energy Environ. Sci. 2014, 7, 4007.
- (27) Lee, J. S.; Won, D. I.; Jung, W. J.; Son, H. J.; Pac, C.; Kang, S. O. Angew. Chem., Int. Ed. 2017, 56, 976.
- (28) Li, F. F.; Lau, J.; Licht, S. Adv. Sci. 2015, 2, 1500260.
- (29) Li, D. W.; Ouyang, S. X.; Xu, H.; Lu, D.; Zhao, M.; Zhang, X. L.; Ye, J. H. Chem. Commun. 2016, 52, 5989.
- (30) Urbain, F.; Tang, P. Y.; Carretero, N. M.; Andreu, T.; Gerling, L. G.; Voz, C.; Arbiol, J.; Morante, J. R. Energy Environ. Sci. 2017, 10, 2256.
- (31) Chu, S.; Fan, S. Z.; Wang, Y. J.; Rossouw, D.; Wang, Y. C.; Botton, G. A.; Mi, Z. Angew. Chem., Int. Ed. 2016, 55, 14260.
- (32) Hinogami, R.; Nakamura, Y.; Yae, S.; Nakato, Y. J. Phys. Chem. B 1998, 102, 974.
- (33) Kong, Q.; Kim, D.; Liu, C.; Yu, Y.; Su, Y.; Li, Y.; Yang, P. D. Nano Lett. 2016, 16, 5675.
- (34) Song, J. T.; Ryoo, H.; Cho, M.; Kim, J.; Kim, J. G.; Chung, S. Y.; Oh, J. Adv. Energy Mater. 2017, 7, 1601103.
- (35) Jang, Y. J.; Jang, J. W.; Lee, J.; Kim, J. H.; Kumagai, H.; Lee, J.; Minegishi, T.; Kubota, J.; Domen, K.; Lee, J. S. Energy Environ. Sci. 2015, 8, 3597.
- (36) Jang, Y. J.; Jeong, I.; Lee, J.; Lee, J.; Ko, M. J.; Lee, J. S. ACS Nano 2016, 10, 6980.
- (37) Yoneyama, H.; Sugimura, K.; Kuwabata, S. J. Electroanal. Chem. Interfacial Electrochem. 1988, 249, 143.
- (38) Torralba-Penalver, E.; Luo, Y.; Compain, J. D.; Chardon-Noblat, S.; Fabre, B. ACS Catal. 2015, 5, 6138.
- (39) Schreier, M.; Luo, J. S.; Gao, P.; Moehl, T.; Mayer, M. T.; Gratzel, M. J. Am. Chem. Soc. 2016, 138, 1938.
- (40) Sahara, G.; Abe, R.; Higashi, M.; Morikawa, T.; Maeda, K.; Ueda, Y.; Ishitani, O. Chem. Commun. 2015, 51, 10722.
- (41) Sahara, G.; Kumagai, H.; Maeda, K.; Kaeffer, N.; Artero, V.; Higashi, M.; Abe, R.; Ishitani, O. J. Am. Chem. Soc. 2016, 138, 14152.
- (42) Hansen, H. A.; Varley, J. B.; Peterson, A. A.; Norskov, J. K. J. Phys. Chem. Lett. 2013, 4, 388.
- (43) Hansen, H. A.; Shi, C.; Lausche, A. C.; Peterson, A. A.; Norskov, J. K. Phys. Chem. Chem. Phys. 2016, 18, 9194.
- (44) Fan, S. Z.; AlOtaibi, B.; Woo, S. Y.; Wang, Y. J.; Botton, G. A.; Mi, Z. Nano Lett. 2015, 15, 2721.
- (45) Rostrup-Nielsen, J. R. Catal. Today 2000, 63, 159.
- (46) Kuhl, K. P.; Hatsukade, T.; Cave, E. R.; Abram, D. N.; Kibsgaard, J.; Jaramillo, T. F. J. Am. Chem. Soc. 2014, 136, 14107.
- (47) Singh, M. R.; Clark, E. L.; Bell, A. T. Phys. Chem. Chem. Phys. 2015, 17, 18924.
- (48) Rosen, J.; Hutchings, G. S.; Lu, Q.; Rivera, S.; Zhou, Y.; Vlachos, D. G.; Jiao, F. ACS Catal. 2015, 5, 4293.
- (49) Kim, S. K.; Zhang, Y.-J.; Bergstrom, H.; Michalsky, R.; Peterson, A. ACS Catal. 2016, 6, 2003.
- (50) Takeda, H.; Ohashi, K.; Sekine, A.; Ishitani, O. J. Am. Chem. Soc. 2016, 138, 4354.
- (51) Hong, D. C.; Tsukakoshi, Y.; Kotani, H.; Ishizuka, T.; Kojima, T. J. Am. Chem. Soc. 2017, 139, 6538.
- (52) Rao, H.; Bonin, J.; Robert, M. ChemSusChem 2017, 10, 4447.

(53) Lian, S. C.; Kodaimati, M. S.; Weiss, E. A. ACS Nano 2018, 12, 568.

- (54) Lin, Y. J.; Kapadia, R.; Yang, J. H.; Zheng, M.; Chen, K.; Hettick, M.; Yin, X. T.; Battaglia, C.; Sharp, I. D.; Ager, J. W.; Javey, A. J. Phys. Chem. C 2015, 119, 2308.
- (55) Qiu, J.; Zeng, G. T.; Ha, M. A.; Ge, M. Y.; Lin, Y. J.; Hettick, M.; Hou, B. Y.; Alexandrova, A. N.; Javey, A.; Cronin, S. B. Nano Lett. 2015, 15, 6177.
- (56) Hepel, M.; Kumarihamy, I.; Zhong, C. Electrochem. Commun. 2006, 8, 1439.
- (57) Guo, X.; Guo, D.; Qiu, X.; Chen, L.; Zhu, W. J. Power Sources 2009, 194, 281.
- (58) Shi, F.; Baker, L. R.; Hervier, A.; Somorjai, G. A.; Komvopoulos, K. Nano Lett. 2013, 13, 4469.
- (59) Chen, Y. W.; Prange, J. D.; Duhnen, S.; Park, Y.; Gunji, M.; Chidsey, C. E. D.; McIntyre, P. C. Nat. Mater. 2011, 10, 539.
- (60) Kim, H. J.; Kearney, K. L.; Le, L. H.; Haber, Z. J.; Rockett, A. A.; Rose, M. J. J. Phys. Chem. C 2016, 120, 25697.
- (61) Kattel, S.; Yan, B. H.; Yang, Y. X.; Chen, J. G. G.; Liu, P. J. Am. Chem. Soc. 2016, 138, 12440.
- (62) Gao, D.; Zhang, Y.; Zhou, Z.; Cai, F.; Zhao, X.; Huang, W.; Li, Y.; Zhu, J.; Liu, P.; Yang, F.; Wang, G.; Bao, X. J. Am. Chem. Soc. 2017, 139, 5652.
- (63) Chang, X. X.; Wang, T.; Gong, J. L. Energy Environ. Sci. 2016, 9, 2177.
- (64) Alvarez, A.; Borges, M.; Corral-Perez, J. J.; Olcina, J. G.; Hu, L. J.; Cornu, D.; Huang, R.; Stoian, D.; Urakawa, A. ChemPhysChem 2017, 18, 3135.
- (65) Graciani, J.; Mudiyanselage, K.; Xu, F.; Baber, A. E.; Evans, J.; Senanayake, S. D.; Stacchiola, D. J.; Liu, P.; Hrbek, J.; Sanz, J. F.; Rodriguez, J. A. Science 2014, 345, 546.
- (66) Rodriguez, J. A.; Liu, P.; Stacchiola, D. J.; Senanayake, S. D.; White, M. G.; Chen, J. G. G. ACS Catal. 2015, 5, 6696.
- (67) Kattel, S.; Yan, B. H.; Chen, J. G. G.; Liu, P. J. Catal. 2016, 343, 115.
- (68) Kattel, S.; Liu, P.; Chen, J. G. G. J. Am. Chem. Soc. 2017, 139, 9739.
- (69) Kattel, S.; Ramirez, P. J.; Chen, J. G.; Rodriguez, J. A.; Liu, P. Science 2017, 355, 1296.
- (70) Baker, L. R.; Kennedy, G.; Van Spronsen, M.; Hervier, A.; Cai, X.; Chen, S.; Wang, L.-W.; Somorjai, G. A. J. Am. Chem. Soc. 2012, 134, 14208.
- (71) Huygh, S.; Bogaerts, A.; Neyts, E. C. J. Phys. Chem. C 2016, 120, 21659.
- (72) Ji, Y. F.; Luo, Y. J. Am. Chem. Soc. 2016, 138, 15896.
- (73) AlOtaibi, B.; Kong, X.; Vanka, S.; Woo, S. Y.; Pofelski, A.; Oudjedi, F.; Fan, S.; Kibria, M. G.; Botton, G. A.; Ji, W.; Guo, H.; Mi, Z. ACS Energy Lett. 2016, 1, 246.
- (74) Bader, R. F. Acc. Chem. Res. 1985, 18, 9.
- (75) Henkelman, G.; Arnaldsson, A.; Jónsson, H. Comput. Mater. Sci. 2006, 36, 354.
- (76) Peterson, A. A.; Abild-Pedersen, F.; Studt, F.; Rossmeisl, J.; Nørskov, J. K. Energy Environ. Sci. 2010, 3, 1311.
- (77) Asadi, M.; Kim, K.; Liu, C.; Addepalli, A. V.; Abbasi, P.; Yasaei, P.; Phillips, P.; Behranginia, A.; Cerrato, J. M.; Haasch, R. Science 2016, 353, 467.
- (78) Li, Q.; Fu, J. J.; Zhu, W. L.; Chen, Z. Z.; Shen, B.; Wu, L. H.; Xi, Z.; Wang, T. Y.; Lu, G.; Zhu, J. J.; Sun, S. H. J. Am. Chem. Soc. 2017, 139, 4290.
- (79) Wellendorff, J.; Lundgaard, K. T.; Møgelhøj, A.; Petzold, V.; Landis, D. D.; Nørskov, J. K.; Bligaard, T.; Jacobsen, K. W. Phys. Rev. B: Condens. Matter Mater. Phys. 2012, 85, 235149.
- (80) Shi, C.; Hansen, H. A.; Lausche, A. C.; Nørskov, J. K. Phys. Chem. Chem. Phys. 2014, 16, 4720.
- (81) Kim, D.; Xie, C. L.; Becknell, N.; Yu, Y.; Karamad, M.; Chan, K.; Crumlin, E. J.; Norskov, J. K.; Yang, P. D. J. Am. Chem. Soc. 2017, 139, 8329.
- (82) Li, X. G.; Bi, W. T.; Chen, M. L.; Sun, Y. X.; Ju, H. X.; Yan, W. S.; Zhu, J. F.; Wu, X. J.; Chu, W. S.; Wu, C. Z.; Xie, Y. J. Am. Chem. Soc. 2017, 139, 14889.
- (83) Li, Y. H.; Xing, J.; Chen, Z. J.; Li, Z.; Tian, F.; Zheng, L. R.; Wang, H. F.; Hu, P.; Zhao, H. J.; Yang, H. G. Nat. Commun. 2013, 4, 2500.
- (84) Tauster, S. J.; Fung, S. C.; Garten, R. L. J. Am. Chem. Soc. 1978, 100, 170.
- (85) Li, Q. Y.; Wang, K.; Zhang, S. L.; Zhang, M.; Yang, H. J.; Jin, Z. S. J. Mol. Catal. A: Chem. 2006, 258, 83.
- (86) Park, J. Y.; Baker, L. R.; Somorjai, G. A. Chem. Rev. 2015, 115, 2781.

<span id="page-8-0"></span>(87) Chen, P. R.; Khetan, A.; Yang, F. K.; Migunov, V.; Weide, P.; Sturmer, S. P.; Guo, P. H.; Kahler, K.; Xia, W.; Mayer, J.; Pitsch, H.; Simon, U.; Muhler, M. ACS Catal. 2017, 7, 1197.

(88) Silvi, B.; Savin, A. Nature 1994, 371, 683.

(89) Vidal, A. B.; Feria, L.; Evans, J.; Takahashi, Y.; Liu, P.; Nakamura, K.; Illas, F.; Rodriguez, J. A. J. Phys. Chem. Lett. 2012, 3, 2275.