

# Cationic-group-functionalized electrocatalysts enable stable acidic CO<sub>2</sub> electrolysis

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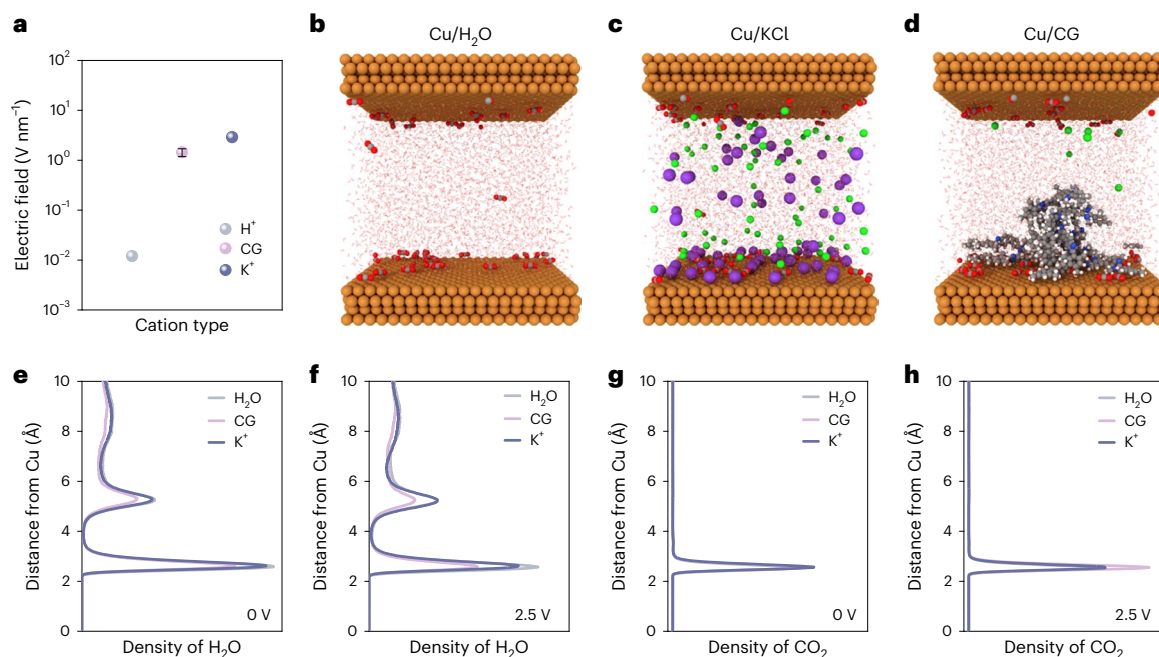
Acidic electrochemical CO<sub>2</sub> reduction (CO<sub>2</sub>R) addresses CO<sub>2</sub> loss and thus mitigates the energy penalties associated with CO<sub>2</sub> recovery; however, acidic CO<sub>2</sub>R suffers low selectivity. One promising remedy—using a high concentration of alkali cations—steers CO<sub>2</sub>R towards multi-carbon (C<sub>2+</sub>) products, but these same alkali cations result in salt formation, limiting operating stability to <15 h. Here we present a copper catalyst functionalized with cationic groups (CG) that enables efficient CO<sub>2</sub> activation in a stable manner. By replacing alkali cations with immobilized benzimidazolium CG within ionomer coatings, we achieve over 150 h of stable CO<sub>2</sub>R in acid. We find the water-management property of CG minimizes proton migration that enables operation at a modest voltage of 3.3 V with mildly alkaline local pH, leading to more energy-efficient CO<sub>2</sub>R with a C<sub>2+</sub> Faradaic efficiency of 80 ± 3%. As a result, we report an energy efficiency of 28% for acidic CO<sub>2</sub>R towards C<sub>2+</sub> products and a single-pass CO<sub>2</sub> conversion efficiency exceeding 70%.

Electrochemical CO<sub>2</sub> reduction (CO<sub>2</sub>R) offers a route to produce fuels and chemicals with low carbon intensity<sup>1–3</sup>. CO<sub>2</sub>R towards multi-carbon (C<sub>2+</sub>) products has reached Faradaic efficiencies (FEs) of 70–80% at relevant current densities (>100 mA cm<sup>-2</sup>) in CO<sub>2</sub>R-favourable alkaline and neutral reaction environments<sup>4–7</sup>. However, at these conditions, the loss of reactant CO<sub>2</sub> to carbonate species limits the single-pass CO<sub>2</sub> conversion efficiency (SPC) (<5%) and thus leads to a significant extra energy cost to regenerate CO<sub>2</sub> from electrolyte<sup>8–11</sup>.

Acidic electrolytes eliminate carbonate formation, enabling high SPC (70–80%) (refs. 12–14); but CO<sub>2</sub>R kinetics are outcompeted by

hydrogen evolution reaction (HER) at low pH<sup>13,15,16</sup>. Adding a high concentration of alkali cations to the electrolyte can help steer the reaction to CO<sub>2</sub>R. However, alkali cations essential for CO<sub>2</sub>R in acidic conditions accumulate, leading to the crystallization of alkali salts on the catalyst and gas diffusion electrode that limit the lifetime of the cell<sup>17,18</sup>. Pulsed electrolysis with a range of duty cycles enables the mitigation of salt formation<sup>17</sup>; however, the CO<sub>2</sub>R performance is compromised by HER due to the local pH change after the open circuit potential (OCP) cycles in acidic conditions. Thus, acidic CO<sub>2</sub>R suffers a trade-off between C<sub>2+</sub> selectivity and stability.

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**Fig. 1** Computational studies of CG-functionalized catalysts in acidic CO<sub>2</sub>R.

**a**, Electric field comparison of H<sup>+</sup>, K<sup>+</sup> and immobilized CG at OHP. The electric field value of CG is average, and the error bar indicates the maximum and minimum electric field values calculated according to the estimated hydrated radius shown in Supplementary Note 2. **b–d**, MD model predictions of atomic configurations for the Cu/H<sub>2</sub>O (**b**), Cu/KCl (**c**) and Cu/CG (**d**) systems at the end of NVE simulation ( $\Delta U = 2.5$  V). Red, grey, yellow, green, purple, blue and

white balls represent O, C, Cu, Cl, K, N and H, respectively. **e, f**, Comparison of simulated density of H<sub>2</sub>O versus distance from the Cu electrode at fixed potential ( $\Delta U$ ) of 0 V (**e**) and 2.5 V (**f**). **g, h**, Comparison of simulated density of CO<sub>2</sub> versus distance from the Cu electrode at fixed potential ( $\Delta U$ ) of 0 V (**g**) and 2.5 V (**h**). Atomic configurations at the start of NVE simulation ( $\Delta U = 0$  V) are shown in Supplementary Fig. 9.  $\Delta U$ , potential difference between two electrodes.

In this Article, we pursued an immobilized cationic group (CG) decorated strategy to improve both the C<sub>2+</sub> selectivity and stability in a strongly acidic environment (0.2 M H<sub>2</sub>SO<sub>4</sub>, pH 0.4). We integrate a thin layer of ionomer with immobilized cationic functional groups at the Cu surface in place of the electrolyte alkali cation. This layer also lowers the proton diffusion rate and increases the local pH for selective C<sub>2+</sub> products. In contrast to the previous acidic work, where a high current density ( $-1$  A cm<sup>-2</sup>) is necessary to overcome HER, we can operate productively at a moderate current density ( $-100$  mA cm<sup>-2</sup>) and thus avoid excessive voltage losses due to cell resistance. When operated at 100 mA cm<sup>-2</sup>, the system maintains 90% SPC of CO<sub>2</sub> and requires a full-cell voltage of 3.3 V. By applying an extra carbon-Nafion layer for uniform ionic current distribution and protection from the direct proton flux, we achieve stable operation (>150 h) and total C<sub>2+</sub> FE of 80%. The strategy presented here provides a catalyst design principle for stable, energy- and carbon-efficient CO<sub>2</sub>R.

## Results

### Alkali cation versus cationic functional group in acidic media

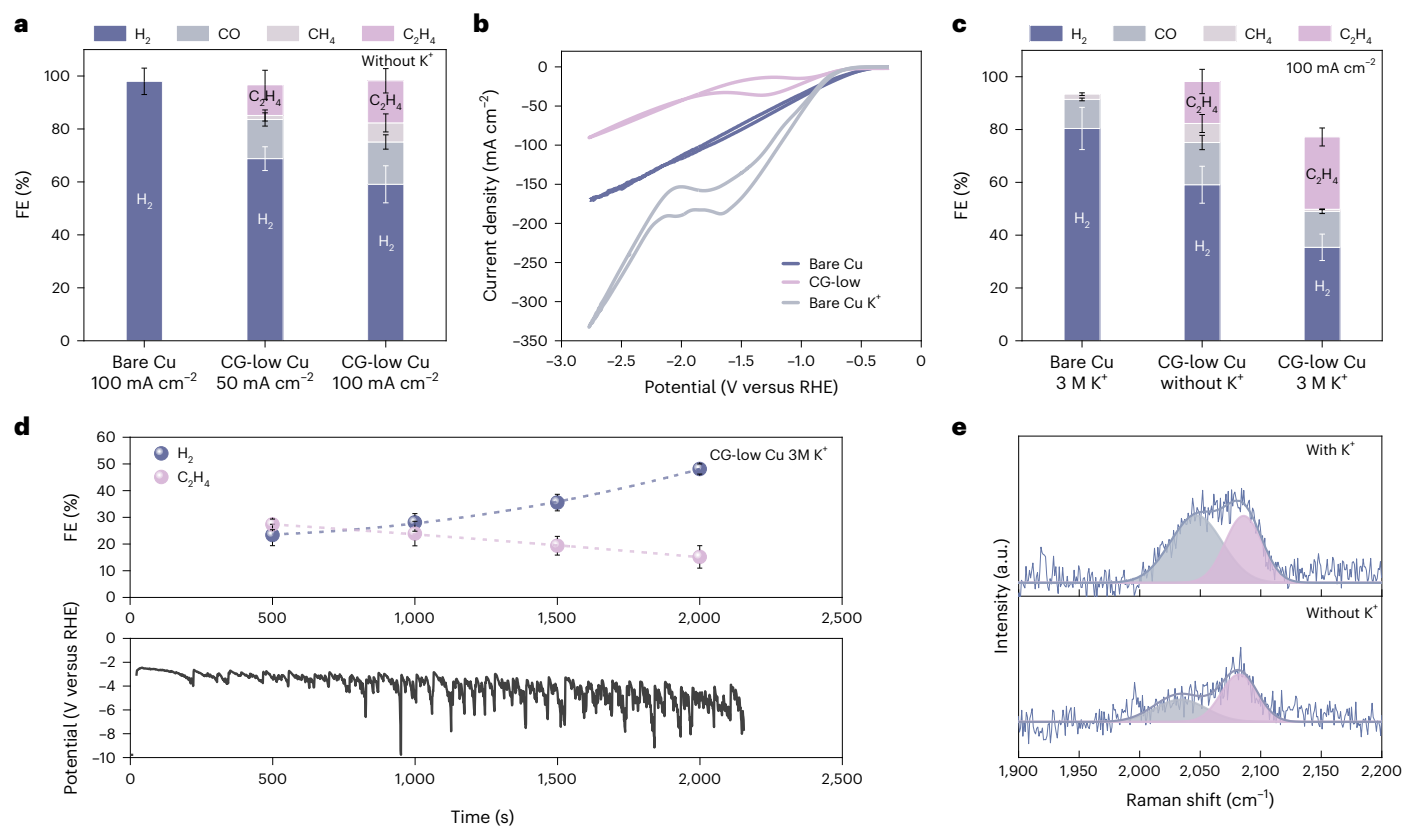
Alkali cations facilitate CO<sub>2</sub>R in strongly acidic environments by modifying the interfacial electric field and buffering local pH<sup>15,19,20</sup> (Supplementary Note 1 and Supplementary Figs. 1 and 2). The high concentration of alkali cations accumulates at cathode surface, leading to salt formation issue (Supplementary Fig. 3 and Supplementary Table 1). We have operated pulsed electrolysis to mitigate the salt issue; however, the CO<sub>2</sub>R selectivity decreases with an increasing HER after OCPs (Supplementary Fig. 4a)—an effect we attribute to reduced alkalinity during the OCP step. The catalyst surface was destroyed by the strong acid when no current or potential was applied during the OCP cycles (Supplementary Fig. 4b).

In addition to the extensive studies on the role of alkali cations in promoting CO<sub>2</sub>R (refs. 13,15,16,21), prior work has suggested that

immobilized CG in ionic liquids or surfactants—including ammonium CG<sup>22</sup>, imidazolium CG<sup>23</sup> and benzimidazole CG<sup>24</sup>—can also stabilize key intermediates and promote CO<sub>2</sub>R to multi-carbon products. These same CG can also bind with functional backbones<sup>25</sup> that potentially influence H<sub>3</sub>O<sup>+</sup> transport.

We predicted the interfacial electric field of the immobilized CG (benzimidazole CG) and the alkali cation (K<sup>+</sup>) using the modified Poisson–Boltzmann model in COMSOL. We found the interfacial electric field generated by CG was of the same order of magnitude as that generated by K<sup>+</sup> (Fig. 1a, Supplementary Note 2 and Supplementary Figs. 5 and 6). The surface electric field generated by cations has a stabilization effect on the negatively charged CO<sub>2</sub> and facilitates CO<sub>2</sub> adsorption and activation in acidic media (Supplementary Fig. 7)<sup>13,15</sup>. In contrast, we found the effect of H<sup>+</sup> on the electric field to be negligible compared with that of both the CG or K<sup>+</sup> ions. We examined the water-management effect via molecular dynamics (MD), quantifying the distribution of H<sub>2</sub>O and CO<sub>2</sub> near the Cu surface. The surface modified with the CG showed the lowest density of H<sub>2</sub>O compared with K<sup>+</sup> and pure water cases at different potentials (Fig. 1b–f, Supplementary Note 3 and Supplementary Figs. 8–15), indicating H<sub>3</sub>O<sup>+</sup>-blocking function. Additionally, the CG layer enhanced the local CO<sub>2</sub> concentration at the Cu surface when a constant potential was applied (Fig. 1b–d, g, h, Supplementary Fig. 15 and Supplementary Videos 1–3)<sup>26</sup>.

Inspired by the cation effect and H<sub>3</sub>O<sup>+</sup> tuning effect predicted by the computational results, we pursued an immobilized cation enrichment strategy deploying a layer of ionomer with the cationic functional groups, one that could vary the concentration of positive benzimidazolium CG on the Cu catalyst (Methods)<sup>27–29</sup>. The number of the CG per dry ionomer is indicated by its ion exchange capacity (IEC). We began with a low concentration of CG (CG-low) that has the lowest water uptake, with which we expected to maximize the local pH and minimize proton migration to the Cu surface.



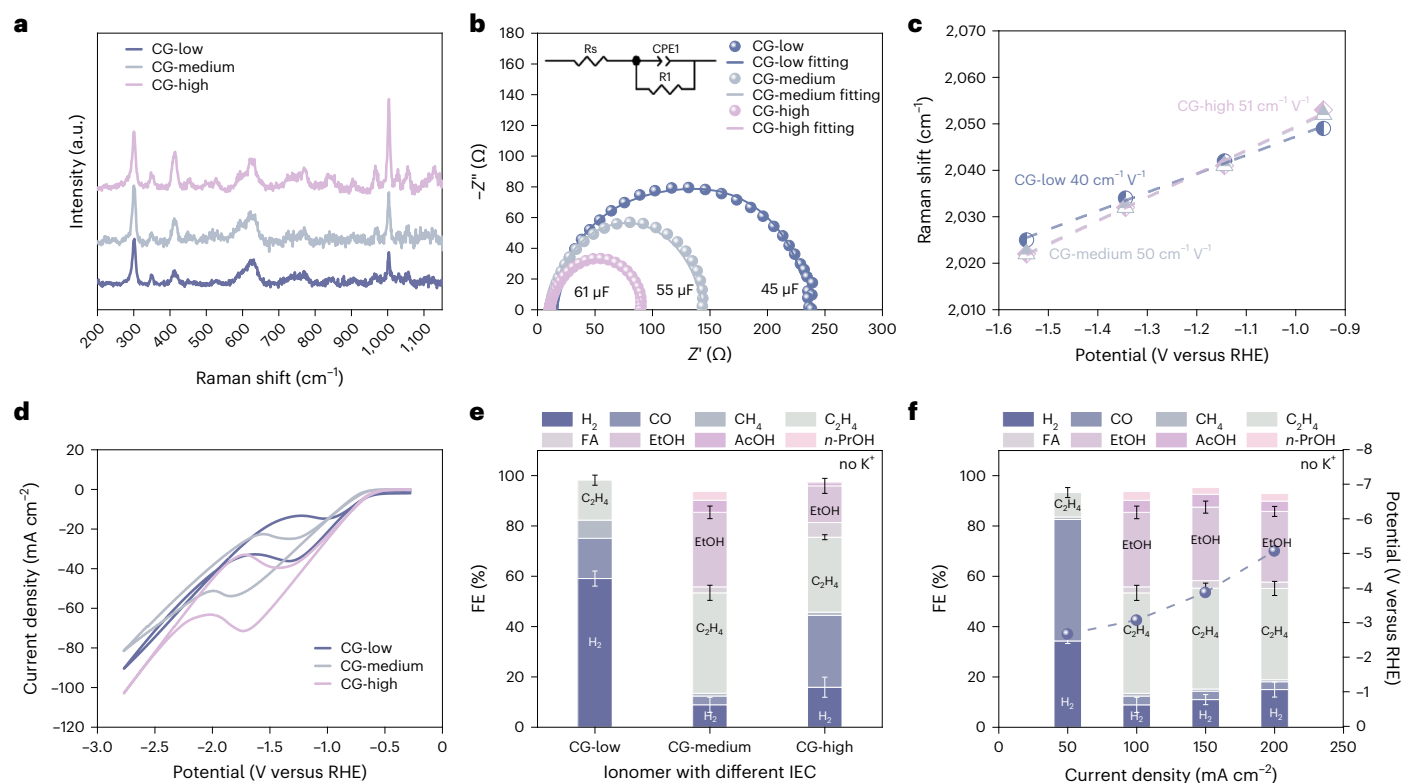
**Fig. 2 | Cationic functional group enables CO<sub>2</sub>R in acidic media. a**, Selectivity of bare Cu at a current density of 100 mA cm<sup>-2</sup> and CG-low Cu at current densities of 50 and 100 mA cm<sup>-2</sup>. Values are means, and error bars indicate s.d. (*n* = 3 replicates). **b**, CV curves of bare Cu (purple), CG-low Cu (pink) in 0.2 M H<sub>2</sub>SO<sub>4</sub> with Ar and bare Cu (grey) in 0.2 M H<sub>2</sub>SO<sub>4</sub> + 3 M KCl with Ar. Scan rate was 100 mV s<sup>-1</sup>. **c**, Selectivity of bare Cu and CG-low Cu in 0.2 M H<sub>2</sub>SO<sub>4</sub> solution with and without K<sup>+</sup> at a current density of 100 mA cm<sup>-2</sup>. Values are means, and error bars indicate s.d. (*n* = 3 replicates). **d**, FE (top) and potential degradation (bottom) of CG-low

Cu at a constant current density of 100 mA cm<sup>-2</sup>. As the catholyte, 0.2 M H<sub>2</sub>SO<sub>4</sub> with 3 M K<sup>+</sup> was used during 30 min. Formation of H<sub>2</sub> bubbles affected the reference electrode and gradually increased noise in the collected potential signals. Values are means, and error bars indicate s.d. (*n* = 3 replicates). **e**, In situ surface-enhanced Raman spectra of CO<sub>2R</sub> on CG-low Cu. The results were obtained at a constant potential -1.34 V versus RHE. The atop-bound CO was deconvoluted into two peaks at HFB (pink) and LFB (grey).

We first examined the CO<sub>2</sub>R performance of CG-low Cu catalyst in a flow cell with a gas diffusion electrode, employing 0.2 M H<sub>2</sub>SO<sub>4</sub> as both catholyte and anolyte. Varying the CG-low loading did not show significant impact on the performance when there was full coverage of the immobilized CG layer (Supplementary Fig. 16). The CG-low Cu showed a reduced H<sub>2</sub> FE of 60%, and CO<sub>2</sub>R products were observed at current densities of 50 and 100 mA cm<sup>-2</sup> (Fig. 2a—an improvement over the negligible amount of CO<sub>2</sub>R (<1%) observed on the bare Cu case at similar electrolyte bulk pH with K<sup>+</sup>). To elucidate the reason of this improvement, we carried out cyclic voltammetry (CV) of CG-low Cu and bare Cu in H<sub>2</sub>SO<sub>4</sub> electrolytes, with and without K<sup>+</sup>. The onset of the plateau that represents the depletion of H<sub>3</sub>O<sup>+</sup> at the Cu surface is at a lower current with CG-low Cu compared with bare Cu in K<sup>+</sup> rich electrolyte (Fig. 2b), indicating a proton barrier effect and higher local pH in the CG-low case, consistent with the prediction of MD simulation results. We then evaluated the electric field and compared it between K<sup>+</sup> and CG-low by performing potential-dependent in situ surface-enhanced Raman spectroscopy (SERS) and calculating the Stark tuning rate<sup>30</sup> (Supplementary Note 4). On CG-low modified Cu, we observed the CO adsorption occupying both terrace and low coordination sites of Cu, similar to that of pure K<sup>+</sup>, but the slightly lower stark tuning rate indicates a smaller surface electric field strength (Supplementary Fig. 17). We posited that CO<sub>2</sub>R activity could be further enhanced with a higher population of CG, as the enhanced electric field can stabilize negatively charged CO<sub>2</sub> intermediates<sup>15,21,31</sup>.

To test this hypothesis, we carried out control studies to examine the performance of CG-low Cu in H<sub>2</sub>SO<sub>4</sub> with the addition of KCl. We detected increased CO<sub>2</sub>R FE, especially towards C<sub>2+</sub> products such as C<sub>2</sub>H<sub>4</sub> (FE 27 ± 2%) at 100 mA cm<sup>-2</sup> with 3 M KCl (Fig. 2c and Supplementary Figs. 18 and 19). This result indicated that the presence of alkali cation accelerates the CO<sub>2</sub>R kinetics on the CG-low Cu surface. The effect of anion species on CO<sub>2</sub>R selectivity was not significant (Supplementary Fig. 20). However, the excess of local OH<sup>-</sup> and K<sup>+</sup> ions led to salt formation at the catalyst and gas channel layers, and a rapid decay in C<sub>2</sub>H<sub>4</sub> FE from 27% to 15% with an increase in H<sub>2</sub> FE from 20% to 48% (Fig. 2d).

To explain the initially lower HER and improvement in C<sub>2+</sub> FE, we measured the CV and in situ SERS in the electrolytes, with and without K<sup>+</sup> on CG-low Cu. With the additional cations, the onset of the current plateau of H<sub>3</sub>O<sup>+</sup> depletion takes place at slightly lower current and overpotential (Supplementary Fig. 21). The additional cation thus further confines the electrical double layer (EDL) and decreases the rate of H<sub>3</sub>O<sup>+</sup> migration towards the outer Helmholtz plane (OHP)<sup>25</sup>. In the absence of the K<sup>+</sup>, the benzimidazolium group is the only cation near the electrode–electrolyte interface. The in situ SERS spectrum showed increasing peak intensities from 1,000 cm<sup>-1</sup> to 1,100 cm<sup>-1</sup>, the frequencies that are associated with the breathing vibration of benzene rings in the benzimidazolium unit, as we ramped up the reductive potential at cathode (Supplementary Fig. 22a). This suggested that the CG (positively charged nitrogen) near the electrode are oriented closer to the negatively charged Cu surface under the increasing



**Fig. 3 | Performance of the CG-functionalized catalysts in acid media.** **a**, Ex situ Raman spectroscopy of CG-low (purple), CG-medium (grey) and CG-high (pink). The characteristic vibration of benzyl group is from 900  $\text{cm}^{-1}$  to 1,100  $\text{cm}^{-1}$ . The peak intensity is normalized by characteristic peak of Cu–O at 300  $\text{cm}^{-1}$ . **b**, EIS measurements for effective EDL capacitance of CG-low, CG-medium and CG-high in 0.2 M  $\text{H}_2\text{SO}_4$ . The EIS data were obtained between 0.1 Hz and 10 kHz. The effective double-layer capacitance ( $C_{dl}$ ) was obtained from the constant phase element (CPE) parameters and the two resistances using the Brug formula. Inset: equivalent circuit for EIS fitting. **c**, Stark tuning slope comparison of CG-

low, CG-medium and CG-high. The atop-bound  $\text{CO}_{ads}$  vibration frequencies in Supplementary Fig. 25 were used to calculate the Stark tuning slope in **c**. **d**, CV curves of bare CG-low, CG-medium and CG-high in 0.2 M  $\text{H}_2\text{SO}_4$  with Ar. Scan rate was 100  $\text{mV s}^{-1}$ . **e**, Selectivity comparison of CG-low, CG-medium and CG-high in 0.2 M  $\text{H}_2\text{SO}_4$  solution at a current density of 100  $\text{mA cm}^{-2}$ . Values are means, and error bars indicate s.d. ( $n = 3$  replicates). **f**, Selectivity comparison of CG-medium Cu at different operating current densities. The tests were carried out in 0.2 M  $\text{H}_2\text{SO}_4$  with a  $\text{CO}_2$  flow rate of 50 sccm. Values are means, and error bars indicate s.d. ( $n = 3$  replicates). FA, formic acid.

**Table 1 | Metrics of ionomers with benzimidazolium cationic functional groups**

Type	IEC <sup>a</sup> ( $\text{mEq g}^{-1}$ )	Conductivity ( $\text{mS cm}^{-1}$ )	Water uptake <sup>b</sup> (%)
CG-low	0.5–0.6	1.3–2.5	13–15
CG-medium	1.4–1.7	2–4	35–50
CG-high	2.3–2.6	8–11	95–100

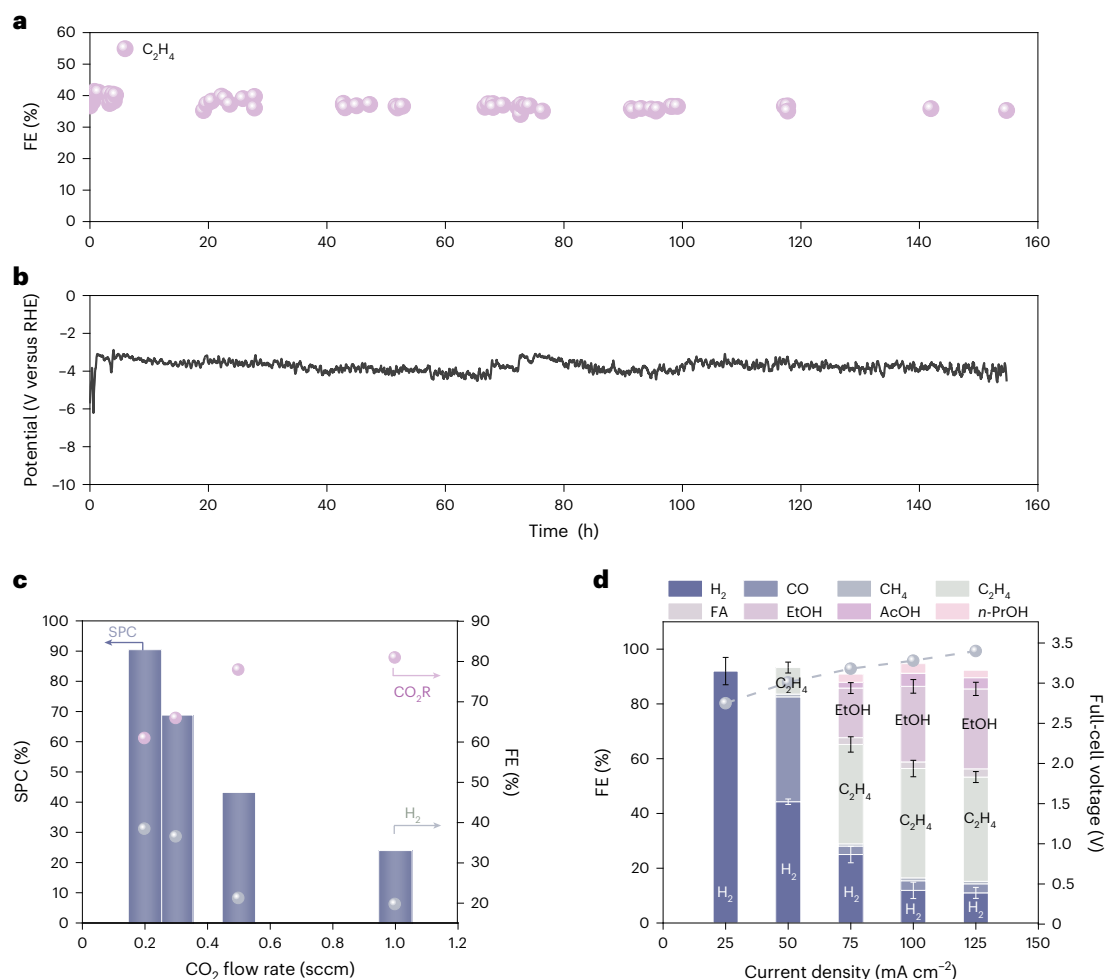
<sup>a</sup>IEC in the hydroxide ( $\text{OH}^-$ ) counter-ion form. <sup>b</sup>Approximate swelling properties when cast into membrane form at 25–50  $\mu\text{m}$ .

electrostatic interactions<sup>32,33</sup>. With the addition of KCl in the electrolyte, we observed an obvious decrease in the peak intensity, indicating a compact cation layer with the binding of hydrated  $\text{K}^+$  that replaced the benzimidazolium group near the Cu surface (Supplementary Fig. 22b). To evaluate the OHP generated by the cation layer, which is correlated with the electric field strength, we quantified the EDL values of CG-low with and without  $\text{K}^+$  addition and compared them with the bare Cu with  $\text{K}^+$  case (Supplementary Fig. 23, Supplementary Note 5 and Supplementary Table 2). The EDL of CG-low without  $\text{K}^+$  addition is smaller than the bare Cu with 3 M  $\text{K}^+$ , in agreement with the trend of Stark tuning slope measured by in situ SERS. Adding  $\text{K}^+$  to the CG-low Cu increases the EDL, indicating a larger OHP that potentially enhances the intermediate adsorption on catalyst surface through enhanced surface electric interactions. Potential-dependent SERS was carried out on CG-low modified Cu in 3 M  $\text{K}^+$  electrolyte, and the Stark tuning

slope was higher than that induced by mere CG-low. This indicates a stronger interfacial electric field near Cu surface in the  $\text{K}^+$  enriched case, consistent with the EDL results (Supplementary Fig. 24). We then compared in situ SERS spectra of Cu–CO in both electrolytes (with and without  $\text{K}^+$ ) to qualitatively assess the cation layer effect on CO adsorption, the main intermediate in  $\text{CO}_2\text{R}$ . When the  $\text{K}^+$  was added into the electrolyte on CG-low modified Cu, at the same electrode potential, we observed a substantial increase in both high-frequency band (HFB) and low-frequency band (LFB) peak area (Fig. 2e), indicating more adsorbed CO on both terrace and low coordination Cu sites and thus fewer sites for  $\text{H}_2$  formation. We also found that, in the presence of  $\text{K}^+$ , the CO occupancy on the terrace site increased more than that on low coordination site (Supplementary Table 3). The improvement in the  $\text{C}_2+$  selectivity is thus attributed to the enhanced surface electric field that significantly increase the CO population, especially on the terrace sites that favour the C–C coupling reaction, albeit at the cost of salt formation and catalyst degradation<sup>21,34,35</sup>.

### Tuning the CG layer in non-alkali electrolyte

To further improve the FE towards  $\text{C}_2+$  products without relying additional alkali cations in solution, we increased the CG concentration. We evaluated the performance of higher CG concentration loaded samples, CG-medium and CG-high modified Cu, at the same condition (pure 0.2 M  $\text{H}_2\text{SO}_4$ ). The increased CG population was confirmed via IEC (Table 1)<sup>36</sup>. The higher CG concentrations were further confirmed through SERS and surface capacitance. The normalized solid-state



**Fig. 4 | Stability and SPC performance of carbon-protected CG-medium Cu in acidic media. a, b**, FE towards C<sub>2</sub>H<sub>4</sub> (**a**) and recorded potential (**b**) at constant current density of 100 mA cm<sup>-2</sup> during 155 h of continuous CO<sub>2</sub>R. **c**, SPC of CO<sub>2</sub> at various flow rates. **d**, Full-cell-voltage performance in a slim flow cell at applied

current densities from 25 mA cm<sup>-2</sup> to 125 mA cm<sup>-2</sup>. Values are means, and error bars indicate s.d. (*n* = 3 replicates). The tests were carried out in the electrolyte of 0.2 M H<sub>2</sub>SO<sub>4</sub>.

spectrum of the CG-medium and CG-high showed higher peak intensity from 1,000 cm<sup>-1</sup> to 1,100 cm<sup>-1</sup>, the frequencies that are associated with the breathing vibration of benzene rings in the benzimidazolium unit (Fig. 3a)<sup>37–39</sup>. There is an observed increase of EDL for CG-medium and CG-high, indicating an enhanced electric field trend when increasing the CG concentration. (Fig. 3b and Supplementary Table 4). Increasing CG concentration (that is, CG-medium and CG-high modified Cu), the CO coverage increased on the basis of the integrated area of  $\nu(\text{CO})$  (Supplementary Fig. 25), which we take as a result of both the increasing amount of available CO<sub>2</sub> molecule near the surface (Supplementary Fig. 15) and faster rate of CO<sub>2</sub> activation due to the increased surface charge density and electric field (Fig. 3c and Supplementary Fig. 25)<sup>13,15</sup>.

Despite the merits of a high CG concentration and high ionic conductivity, high IEC will also lead to enhanced water uptake of the CG layer and greater proton access<sup>40</sup> (Table 1 and Supplementary Fig. 26). In acidic electrolyte, this results in rapid proton influx and a lower local pH unfavourable for CO<sub>2</sub>R. To test this hypothesis, we first measured the CV curve of three different concentration CG samples (CG-low, CG-medium and CG-high) in pure H<sub>2</sub>SO<sub>4</sub> electrolyte (0.2 M) to investigate the proton depletion current. As expected, the plateau current of HER increased with the increasing IEC and water uptake, indicating a faster H<sub>3</sub>O<sup>+</sup> migration from the bulk electrolyte when a higher CG concentration layer is applied (Fig. 3d). Although a higher

CG concentration is desired at OHP, the enhanced water uptake compromises the effectiveness of the CG layer as proton barrier<sup>41,42</sup>. It is therefore important to tune both the CG density and water/proton permeability to reach the highest C<sub>2+</sub> FE.

We then tested the performance of CG-modified Cu at a range of acid concentrations (Supplementary Fig. 27). The best result sustained over a 1 h test was that of CG-medium Cu, with an FE of 40 ± 2% towards C<sub>2</sub>H<sub>4</sub>, much lower HER (<10%) and a total C<sub>2+</sub> FE of 80 ± 3% at current density of 100 mA cm<sup>-2</sup> (Fig. 3e, f and Supplementary Fig. 28). CG-high resulted in more CO detected (28 ± 2%) and less total C<sub>2+</sub> products to 60 ± 4%. The slightly higher CO FE and less C<sub>2</sub>H<sub>4</sub> FE on CG-high modified Cu, which is also consistent with the lower CO<sub>ads</sub> terrace ratio observed from in situ SERS (Supplementary Fig. 25 and Supplementary Table 5), was attributed to the higher water uptake that increases the proton transfer from bulk to the Cu surface and thus lowers the local pH (Fig. 3d). To quantify its value and directly compare the local pH of CG-medium and CG-high, we used 0.5 M H<sub>3</sub>PO<sub>4</sub> solution as a pH indicator during in situ SERS measurement. We recorded the signal areas of H<sub>3</sub>PO<sub>4</sub>, H<sub>2</sub>PO<sub>4</sub><sup>-</sup>, HPO<sub>4</sub><sup>2-</sup> and PO<sub>4</sub><sup>3-</sup> and used the equilibrium of H<sub>3</sub>PO<sub>4</sub>/H<sub>2</sub>PO<sub>4</sub><sup>-</sup>/HPO<sub>4</sub><sup>2-</sup>/PO<sub>4</sub><sup>3-</sup> to calculate the local pH (Supplementary Note 6 and Supplementary Table 6). At 150 mA cm<sup>-2</sup> operating current density, the local pH of CG-high Cu is calculated to be -12.6, slightly lower than that of CG-medium Cu's -12.9 (Supplementary

Fig. 29 and Supplementary Table 7). The lower local pH leads to a kinetic competition between CO desorption and C–C coupling that reduces C<sub>2+</sub> selectivity (Supplementary Figs. 30 and 31)<sup>43,44</sup>. Adding more K<sup>+</sup> sources into the electrolyte enhanced the EDL structure of CG-medium and CG-high, but the local pH was affected by the increasing water permeability introduced by the hydrated K<sup>+</sup> (Supplementary Note 7, Supplementary Figs. 32–35 and Supplementary Table 8). Similar to the CG-low case, the additional K<sup>+</sup> containing electrolytes resulted in severe salt precipitation and rapid performance degradation (Supplementary Fig. 36)

In non-alkali acidic electrolyte, CG-medium creates a productive catalytic condition that balances cation group density and water uptake. Concentrated immobilized non-alkali CG enable stable CO<sub>2</sub>R performance with H<sub>2</sub> FE less than 10% and C<sub>2+</sub> FE around 80%. The benefit brought by further increasing CG concentration is limited and will be compromised by the higher water uptake that leads to faster H<sub>3</sub>O<sup>+</sup> migration.

### Acidic CO<sub>2</sub>R performance with the CG-modified catalyst

Replacing the alkali metal cation in the solution with a layer of CG on Cu should reduce salt formation and thereby remove a leading cause of poor system stability. However, CG-medium Cu (in 0.2 M H<sub>2</sub>SO<sub>4</sub>) provided C<sub>2</sub>H<sub>4</sub> FE >35% for less than 1 h (Supplementary Fig. 37). We observed dark spots formed on the Cu (Supplementary Fig. 38), which we attribute to the corrosion of Cu as the protective cation group layer decayed over time, followed by an uneven distribution of current density. The hydration of CG over the layer proceeds at different rates, leading to non-uniformities in hydration, ionic conductivity and access for protons to oxidize the Cu underneath. We applied a layer of carbon nanoparticles (CNPs) atop the CG layer to serve as a physical barrier to limit the proton transfer. The carbon layer may also provide the electrode some protection from electrolyte impurities detrimental to CO<sub>2</sub>R performance and stability<sup>45</sup> (Supplementary Table 9). This layer was made up of CNPs bound with Nafion to prevent breakdown of the CG layer due to non-uniform hydration and ionic conductivity. This protective layer reduces the proton flux through the CG layer by enhancing the surface hydrophobicity and does not substantially change its EDL structure (Supplementary Figs. 39 and 40 and Supplementary Table 10). The Nafion/carbon-protected CG-medium Cu operated for over 150 h in strong acid, with a steady C<sub>2</sub>H<sub>4</sub> FE of over 40% (Fig. 4a,b). The carbon-protected CG layer remained intact after >150 h operation (Supplementary Fig. 41). Only trace K<sup>+</sup> was detected in the electrolyte after 80 h of electrolysis (Supplementary Table 11).

The lack of salt formation in this system also benefits CO<sub>2</sub> utilization. CO<sub>2</sub> is regenerated within the bulk electrolyte and fed back to the gas inlet. We pursued a high SPC to reduce further the product separation costs. SPC is calculated using the fraction of the input CO<sub>2</sub> supply that is converted to CO<sub>2</sub>R products. By throttling the input CO<sub>2</sub>, we achieved 90% SPC for all CO<sub>2</sub>R products (Fig. 4c). Further employing a slim flow cell with low resistance (Supplementary Fig. 42), we operated the carbon-coated CG-medium Cu at current densities from 50 to 150 mA cm<sup>-2</sup>, and the full-cell voltage at 100 mA cm<sup>-2</sup> was 3.3 V (without iR compensation) with a C<sub>2+</sub> FE of 80%, leading to a record C<sub>2+</sub> energy efficiency (EE) of 28% for acid CO<sub>2</sub>R system (Fig. 4d). We assessed pure water electrolysis in a zero-gap system using Nafion as the proton conductor. The FE towards C<sub>2</sub>H<sub>4</sub> was similar to that in acidic electrolyte, but the high ionic resistance of pure water led to high full-cell voltage and low EE (Supplementary Fig. 43).

## Conclusions

This work demonstrates that CO<sub>2</sub>R in an acidic environment can be achieved without alkali metal cations in solution and the associated challenges of salt formation and CO<sub>2</sub> loss. Immobilized cationic functional groups generate a surface electric field and provide a buffer of local acidity that enables CO<sub>2</sub> activation and C–C coupling in the

presence of strong acid. We demonstrate that the local environment can be made productive for C<sub>2+</sub>, without requiring high current densities that incur severe energy costs. Operating at a moderate current density of 100 mA cm<sup>-2</sup>, we achieve 80% C<sub>2+</sub> FE and 9% H<sub>2</sub> FE in a strong acid environment (pH 0.4) with no alkali metal cations in solution. This approach requires a full-cell potential of 3.3 V (iR-free) to operate CO<sub>2</sub>R at an industrially relevant reaction rate of 100 mA cm<sup>-2</sup>. By avoiding carbonate salt formation from alkali metal ions, the stability of this acidic system exceeds 150 h. Additionally, loss of CO<sub>2</sub> is minimized, and the maximum SPC exceeds 90%. This study highlights a strategy to design CO<sub>2</sub>R catalysts in non-alkali acidic electrolytes that eliminates the energy cost related to salt formation and enables prolonged CO<sub>2</sub>R in acidic conditions.

## Methods

### Preparation of Cu electrodes

The Cu electrode was prepared by sputtering pure Cu (>99.99%, Kurt J. Lesker) onto a polytetrafluoroethylene gas diffusion layer with 450 nm pore size. The sputtering was carried out in an Angstrom Nexdep sputtering system in a vacuum environment (–10<sup>-5</sup>–10<sup>-6</sup> torr) with a deposition rate of 1 Å s<sup>-1</sup>. The thickness of the sputtered Cu layer was 200 nm.

### Preparation of CG-modified Cu electrodes

The CG-modified Cu electrodes were fabricated by spraying 1 wt% ionomer solutions with different cationic functional group densities and the spray loadings were kept constant at 3 mg cm<sup>-2</sup>.

All three CG-low, CG-medium and CG-high ionomers were prepared by dissolving three types of ionomer dry resin (Aemion<sup>+</sup>, ionomer AP2-HNN2-00, AP2-HNN5-00 and AP2-HNN8-00, respectively) in a solvent composed of 80/20 vol.% ethanol (>99.5%, Sigma-Aldrich)/acetone (>99.5%, Fisher Chemical) to yield a 1 wt% ionomer solution. The spray loading of the ionomer solution was adjusted from 1 to 3 mg cm<sup>-2</sup>. Carbon-protected CG-Cu was prepared by spray coating CNP ink onto the CG-modified Cu. The CNP ink was composed of 4 mg CNP (Vulcan XC 72R) and 0.035 g Nafion solution (5 wt%, D520 Dispersion) dispersed in 15 ml methanol (>99.8%, Fisher Chemical). The CNP ink was sonicated for at least 1 h before spray coating.

### Electrochemical reduction of CO<sub>2</sub>

The CO<sub>2</sub>R was carried out in a three-electrode flow cell, where CG-modified Cu as the cathode electrode with an exposed size of 1 cm<sup>2</sup>, an Ag/AgCl (3 M KCl) as the reference electrode and a platinum gauze (99.99%, Sigma-Aldrich) as the counter electrode. The H<sub>2</sub>SO<sub>4</sub> catholyte and anolyte (>95%, ACS reagent, Sigma-Aldrich), with concentrations varied from 0.01 M to 0.2 M, were circulated in the flow cell at a constant flow rate of 10 ml min<sup>-1</sup>. The catholyte and anolyte were separated by a proton exchange membrane (Nafion 117). The CO<sub>2</sub> was supplied at a flow rate of 50 standard cubic centimetres per minute (sccm) by using a digital mass flow controller. All the electrochemical tests were performed through a potentiostat (Autolab PGSTAT302N). The volumes of catholyte and anolyte used for circulation were 25 ml, and the liquid products were collected after 1 h of continuous operation for analysis. Linear sweeping voltammetry measurements were carried out in the same flow cell at a scan rate of 50 mV s<sup>-1</sup>. All potentials were converted to reversible hydrogen electrode (RHE) via the following equation:

$$E(\text{RHE}) = E(\text{Ag/AgCl}) + 0.210 \text{ V} + 0.059 \times \text{pH} \quad (1)$$

The ohmic resistance and charge transfer resistance were measured through electrochemical impedance spectroscopy (EIS), and the data points were obtained between 0.01 Hz and 200 kHz.

### CO<sub>2</sub>R product analysis

The gas products were analysed through gas chromatograph (Perkin Elmer Clarus 590) equipped with a thermal conductivity detector and

flame ionization detector. The gas products were controlled in 1 ml volume from the gas outlet and injected into the gas chromatograph for quantification, and the FE was calculated via the following equation:

$$\text{Faradaic efficiency (\%)} = \frac{zFP}{RT} \times \nu \times \frac{1}{I} \times 100\% \quad (2)$$

where  $z$  represents the number of electrons required to produce the product,  $F$  represents the Faraday constant,  $P$  represents the atmosphere pressure,  $R$  represents the ideal gas constant,  $T$  represents the temperature,  $\nu$  represents the gas flow rate of the gas and  $I$  represents the total current.

The liquid products were analysed using proton nuclear magnetic resonance spectroscopy (600 MHz Agilent DD2 NMR Spectrometer) under water suppression mode. Dimethyl sulfoxide was used as the reference and deuterium oxide as the lock solvent. The FE of liquid was calculated via the following equation:

$$\text{Faradaic efficiency (\%)} = \frac{znF}{It} \times 100\% \quad (3)$$

where  $z$  represents the number of electrons required to produce the product,  $n$  represents the mole number of products and  $t$  represents the electrolysis time.

The SPC of CO<sub>2</sub> was calculated using the following equation<sup>12</sup>:

$$\text{SPC (\%)} = \frac{\frac{j}{zF} \times V_m}{\text{Flow rate}} \times 100\% \quad (4)$$

where  $j$  represents the partial current density of a specific product and  $V_m$  represents the molar volume.

### Materials characterization

Scanning electron microscopy cross-section images were obtained in a high-resolution scanning electron microscope (Hitachi S-5200). In situ Raman measurements were obtained using a Renishaw inVia Raman microscope equipped with a water immersion objective (63×) with a 785 nm laser in a modified flow cell. Considering that at current densities greater than 100 mA cm<sup>-2</sup>, H<sub>2</sub> bubbles generated from Cu will cover the lens and deteriorate the quality of Raman signals, we used 5 mM H<sub>2</sub>SO<sub>4</sub> for all the tests and applied a constant current density of 20 mA cm<sup>-2</sup> for all the in situ measurements. For the potassium-rich cases, 3 M KCl was added in the 5 mM H<sub>2</sub>SO<sub>4</sub> electrolyte. CO<sub>2</sub> was supplied to the cathode during all the in situ measurements.

### Slim flow cell configuration

The cathodes for the slim flow cell were the CG-medium Cu electrodes with a CG-medium ionomer loading of 3 mg cm<sup>-2</sup>. The anodes were made by IrO<sub>2</sub>-coated Ti felt (0.3 mm thickness) with a loading of 1 mg cm<sup>-2</sup> (refs. 46,47). The measurements were performed in a slim flow cell with an active area of 1 cm<sup>2</sup> accessed with a serpentine channel. The catholyte was circulated in a flow field with a thickness of 0.4 mm as shown in Supplementary Fig. 42. CO<sub>2</sub> was fed into the cathode at a flow rate of 50 sccm by using an accurate mass flow controller. The 0.2 M H<sub>2</sub>SO<sub>4</sub> catholyte and anolyte were circulated at a constant flow rate of 5 ml min<sup>-1</sup> using peristaltic pumps. A cation exchange membrane (Nafion 117, Fuel Cell Store) was used for ion exchange and separation of the cathode and anode.

### Multiphysics simulations

The simulation of space charge density and electric field in 1D domain (Supplementary Fig. 5) were performed in the COMSOL Multiphysics package based on the finite-element-method solver (<https://www.comsol.com/>). The electric double layer at the electrode–electrolyte interface was modelled using the Gouy–Chapman–Stern model<sup>48</sup>,

which consists of the Stern layer and the diffuse layer. The Stern layer consists of a monolayer of surface-adsorbed hydrated cations on the electrode surface, with a thickness of  $d_s$  depending on the hydrated cation. The diffuse layer contains free anions and cations distributed according to the Poisson and Boltzmann law, which forms the concentration gradient away from the electrode surface.

### MD simulation

Classical MD simulations of full cells of the Cu/KCl/CG/CO<sub>2</sub> solution interface were performed using LAMMPS<sup>49</sup>. The initial structures of interfaces were constructed using PACKMOL (Supplementary Fig. 9)<sup>50</sup>. Each electrode is composed of four copper (111) layers with a surface area of 51.12 × 48.70 Å<sup>2</sup> (3,520 Cu atoms in total). The left and right electrodes are separated by a vacuum region of -12.5 Å, and the total cell length ( $z$  coordinate) is 70.26 Å. The electrolyte initially consists of -3,600 H<sub>2</sub>O molecules and 40 randomly distributed CO<sub>2</sub> molecules (Supplementary Fig. 9a). For Cu/KCl solution, 65 KCl ion pairs (corresponding to a 1 M concentration) were added (Supplementary Fig. 9b). For Cu/CG solution, a C<sub>50</sub>H<sub>51</sub>N<sub>4</sub> molecule with +1 charge is used as the building block for the ionomer (Supplementary Fig. 9c). The initial structure of C<sub>50</sub>H<sub>51</sub>N<sub>4</sub> molecule was optimized by Gaussian 16 (ref. 51) B3LYP/6-311 G(d, p) (ref. 52). Considering the size of our model, eight C<sub>50</sub>H<sub>51</sub>N<sub>4</sub> molecules are added into the solution to model Cu/CG solution and same number of Cl<sup>-</sup> ions are added to balance the charge.

The pairwise interactions between atoms were described using Lennard–Jones potential, and the long-range Coulomb interactions were calculated using the particle–particle particle–mesh method with an accuracy of 10<sup>-5</sup> with a cut-off distance of 12 Å. The SPC/E water model was used to reproduce the aqueous environment, with the SHAKE algorithm to constrain the stretching mode of water. The 12–6 Lennard–Jones potential parameters of Cu were obtained from ref. 53, for KCl from ref. 54 and for CO<sub>2</sub> from ref. 55. Parameters for the ionomer were generated with the Moltemplate<sup>56</sup> using the second generation of general Amber force field (GAFF2) (ref. 57), partial charges were set to fit the electrostatic potential generated with B3LYP/6-31G\*\* by RESP<sup>58</sup> using Gaussian 16 (ref. 51) (Fig. 1b–d). The parameters for the cross atoms were obtained using the Lorentz–Berthelot mixing rules.

The simulations were first performed in NPT ensemble in the 2D-PBC model at 298.15 K and 1 bar. The time step was set to 1 fs and atom positions were updated with Verlet algorithm. The interface was equilibrated by at least 10 ns. The equilibrated interface model was then transferred to a 3D-PBC set-up, in which the electrode potentials can be adjusted using the finite field method implemented in the USER-CONP2 package<sup>59</sup>. The package, described in ref. 59, has been used in simulating electric double layers<sup>60–62</sup>. With this package, we adjusted electrode potentials using the finite field method at fixed potential differences ( $\Delta U$ ) of 0.0, 0.5, 1.0, 1.5, 2.0 and 2.5 V between two electrodes.

The simulations were performed in NVE ensemble at 298.15 K for at least 30 ns. The temperature is adjusted with a canonical sampling thermostat that uses global velocity rescaling with Hamiltonian dynamics<sup>63</sup>. The final 10 ns trajectories were used for post-analysis with MDAnalysis Python package<sup>64</sup> and visualized by OVITO<sup>65</sup>.

### Data availability

Source data for the stability test shown in Fig. 4a,b and the atomic coordinates of the optimized computational models can be found in figshare<sup>26</sup>. Data that support the findings of this study can be found in the article and Supplementary information. All other data supporting this work are available from the corresponding authors upon reasonable request.

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## Author contributions

D.S. and E.H.S. supervised the project. M.F., J.E.H. and R.K.M. designed and carried out all the experiments. J.E.H. conceived the idea. Y.M., P.O. and Y.C. carried out the MD simulation. Z.W. supervised the MD simulation. Y.M., Y.C., G.I.N.W. and Z.W. analysed the MD simulation results. M.F. and J.E.H. analysed the experimental data and prepared the paper. R.K.M. performed the slim-flow cell design and experiments. F.L. carried out the COMSOL simulation. M.F. carried out all EIS measurements and analysed data. J.E.H. and Z.C. carried out the SERS measurements, and M.F. analysed the SERS data. Z.Z. performed the pH-sensitive Raman tests. J.Z., A.O. and Y.W. synthesized catalysts. W.N. and Y.Z. carried out scanning electron microscopy characterization. Y.Y. performed the X-ray diffraction characterization. B.K. and K.G. carried out the contact angle measurements. C.P.O., Y.X. and Y.C.X. performed product analysis. All authors discussed the results and assisted during paper preparation.

## Competing interests

There is a US provisional patent application (63/381180) titled 'A modified catalyst for operating electrochemical carbon dioxide reduction in a non-alkali acidic medium and related techniques', filed by the authors M.F., J.E.H., R.K.M., E.H.S. and D.S. of this article and their institutions. The other authors declare no competing interests.

## Additional information

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